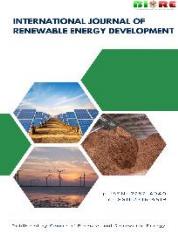




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Research Article

Comparative study of $\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$ and $\text{BiVO}_4/\text{Cu}_2\text{O}$ photocathodes for enhanced electricity generation and hydrogen evolution in photocatalytic fuel cells

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Abstract. The preparation of good photocathodes is a crucial issue regarding promoting the performance of photocatalytic fuel cell (PFC) systems for environmentally protective energy conversion approaches. In the present work, a comparative study of Cu_2O -based photocathodes jointly modified with graphitic carbon nitride ($\text{g-C}_3\text{N}_4$) and bismuth vanadate (BiVO_4) was performed to ascertain their competence towards concomitant electricity generation and hydrogen evolution in an integrated single-chamber photocatalytic fuel cell. Cu substrates were anodized to produce ordered Cu_2O layers, modified with immersion treatments, and then low-temperature calcination. The as-prepared products were characterized in detail by XRD, HR-TEM, UV-Vis DRS, PL spectra, and XPS analyses, as well as photoelectrochemical measurements to gain insight into crystallinity, morphology, photocatalytic activity (optical absorption), electronic structure, and charge recombination. Results revealed that among the pristine Cu_2O and $\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$, superior charge separation was exhibited on the $\text{BiVO}_4/\text{Cu}_2\text{O}$ photocathode, along with better power density and hydrogen evolution. The highest power density of $\text{BiVO}_4/\text{Cu}_2\text{O}$ was $0.05625 \text{ mW cm}^{-2}$ and $13.71 \text{ mmol.m}^{-2}$ for hydrogen evolution compared to both Cu_2O ($0.0375 \text{ mW cm}^{-2}$; $11.19 \text{ mmol.m}^{-2}$) and $\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$ (0.026 mW cm^{-2} ; $8.1616 \text{ mmol.m}^{-2}$). This observation was supported by the analysis of the photoluminescence spectra: $\text{BiVO}_4/\text{Cu}_2\text{O}$ had PL intensity of 325 a.u., lower than Cu_2O (400 a.u.) and $\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$ (650 a.u.), validating that this sample more effectively suppressed electron-hole recombination and electron transport mechanism. The enhanced photocatalytic activity of $\text{BiVO}_4/\text{Cu}_2\text{O}$ is associated with the generation of a p-n heterojunction, which accumulates a built-in electric field to drive effective charge separation and offers visible-light sensitization upon its larger absorption spectrum that is beneficial for not only promoting hydrogen evolution efficiency but also improving electricity production in PFC systems.

Keywords: Cu_2O , photocathode, hydrogen evolution, photocatalytic fuel cell, Z-scheme.



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1. Introduction

The growing demand for renewable energy and stricter environmental regulations has increased the need for clean and decentralized energy systems. Hydrogen has garnered a lot of interest as a possible fuel for the energy infrastructure of the future because of its carbon-free qualities, high energy density, and portability among green energy carriers. One intriguing solar-driven device that can generate both power and hydrogen at the same time is the photocatalytic fuel cell (PFC). (Liu *et al.*, 2024; Rehan *et al.*, 2025). The photoelectrode, and the photocathode in particular, are crucial to the operation of PFCs since they are the ones that really carry out the hydrogen evolution process (HER). (He *et al.*, 2019). Cu_2O , or copper(I) oxide, is a fascinating material for use as a photocathode since it is a p-type semiconductor with a suitable bandgap (~2.2 eV) and an advantageous conduction band edge for HER. (Hamdani & Bhaskarwar, 2021) (N. Li *et al.*, 2024). However, Cu_2O

photocatalytic efficacy is severely limited due to its electron-hole recombination and weak photostability. Surface modification and semiconductor coupling, two methods of heterojunction engineering, have been extensively investigated as potential solutions to these limitations (Lu *et al.*, 2025).

The 2D polymeric semiconductor graphitic carbon nitride ($\text{g-C}_3\text{N}_4$) has a bandgap of 2.7 eV, excellent visible light absorption capabilities, and excellent thermal and chemical durability. The formation of a p-n heterojunction by $\text{g-C}_3\text{N}_4$ complexed with Cu_2O has many benefits, including enhanced photocatalytic activity, longer lifespan, photoelectron-hole recombination, and charge separation and adsorption (Mamari *et al.*, 2021; Dai *et al.*, 2022).

Conversely, in the type-II or Z-scheme structure, the n-type semiconductor bismuth vanadate (BiVO_4) serves as an effective electron mediator due to its visible area and about 2.4 eV band gap. Incorporating BiVO_4 enhances the photocathode

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surface's reduction capacity, stability, and photocurrent production capabilities. (Dai *et al.*, 2022; Lin *et al.*, 2014; Cheng *et al.*, 2025).

Their heterojunction systems have been extensively researched, but their use as photocathodes for PFCs, particularly for the evolution of both hydrogen and electricity at the same time, has been reported only seldom. Therefore, according (Sienkiewicz *et al.*, 2021), a mix of anodization and immersion-calcination methods was used to create Cu_2O photocathodes modified with $\text{g-C}_3\text{N}_4$ or BiVO_4 . A single-chamber PFC was used to test the electrodes, and the electrolyte was made from POME, which is a byproduct of the palm oil mill. A thorough assessment using structural, optical, and photoelectrochemical studies showed how each alteration affected charge transfer, photostability, and overall photocathodic performance.

The novelty of this study lies in two key aspects. First, this work employs a flat Cu metal plate as the conductive substrate, which is directly converted into Cu_2O and further modified into $\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$ and $\text{BiVO}_4/\text{Cu}_2\text{O}$ heterojunctions. This approach provides shorter electron pathways, stronger interfacial bonding, and improved mechanical stability compared with conventional FTO, ITO, carbon cloth, or Cu foam-based photocathodes. Second, both heterojunction photocathodes are integrated into a single-chamber PFC combining hydrogen evolution and electricity generation, enabling one of the first systematic evaluations of two distinct Cu_2O heterostructures within the same operating configuration.

2. Materials and Methods

2.1 Materials and Reagents

There were no further purifying procedures performed on any of the compounds used in this investigation; they were all of the pro-analysis grade. The photocathode was manufactured using a purified copper (Cu) plate with a thickness of 0.3 mm and a purity of 99.9% or higher. We got ammonium metavanadate (NH_4VO_3), bismuth pentahydrate nitrate ($\text{Bi}(\text{NO}_3)_3 \cdot 5\text{H}_2\text{O}$), melamine, and potassium hydroxide (KOH) from Merck. During synthesis, ethanol and deionized water were used for all cleaning and dissolving operations.

2.2 Synthesis of Cu_2O -based Photocathodes

Two sorts of photocathodes, $\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$ and $\text{BiVO}_4/\text{Cu}_2\text{O}$, were fabricated respectively. The Cu substrates were transformed to Cu_2O by anodization and surface modified using immersion and calcination methods, respectively.

2.2.1 Formation of Cu_2O Layer via Anodization

After being ultrasonically cleaned in 0.25 M HCl, acetone, ethanol, and deionized water in sequence, the 8x4 cm copper plates were dried. In a 2 M KOH solution, electrolyte anodization was carried out for 20 minutes at room temperature with a voltage of 1.2 V (Bachri *et al.*, 2025).

2.2.2 Modification with $\text{g-C}_3\text{N}_4$

Graphitic "carbon nitride" ($\text{g-C}_3\text{N}_4$) was prepared thermally by the heating of melamine powder in a closed crucible at 550 °C for 2 hours in a furnace, and it appeared as a yellow powder (Husein *et al.*, 2024). For the preparation of $\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$ composite, the anodized Cu_2O was ultrasonicated in the $\text{g-C}_3\text{N}_4$ suspension, which was synthesized by ultrasonication of 0.5 g $\text{g-C}_3\text{N}_4$ in isopropyl alcohol for 5 min and

then was left to stand for 15 h. The samples were sintered at 350 °C for 1 hour under nitrogen with a heating rate of 5 °C min⁻¹.

2.2.3 Modification with BiVO_4

The $\text{BiVO}_4/\text{Cu}_2\text{O}$ composite was made by burning the anodized Cu_2O plates for one hour at 350°C in a nitrogen environment at a rate of 5°C/min. The BiVO_4 precursor solution was prepared as follows: First, 100 mL of deionized water was used to dissolve 0.485 g of $\text{Bi}(\text{NO}_3)_3 \cdot 5\text{H}_2\text{O}$ (Solution A). Then, 100 mL of hot deionized water at 70 °C was used to dissolve 0.117 g of NH_4VO_3 (Solution B). Ten minutes were spent stirring the two liquids together. Then, the Cu_2O electrodes were immersed in the mixture for 3 h (optimized duration). These conditions were selected based on preliminary trials to ensure uniform coating, good interfacial contact, and phase stability of both Cu_2O and the deposited layer without inducing over-oxidation or structural degradation.. the Cu_2O electrodes were removed and allowed to dry naturally. The last step was to calcine the electrode in air at 200 °C for an hour.

2.3 Characterization Techniques

The crystalline phase and the microstructure of the samples were identified using X-ray diffraction (XRD, PANalytical X'Pert PRO, Cu K α radiation), transmission electron microscopy (TEM, JEOL JEM-2100), and X-ray photoelectron spectroscopy (XPS, PHI 5000 VersaProbe).

The UV-vis spectrum of the diffuse reflectance was measured from a Shimadzu UV-2600 recording spectrophotometer, from which the band gap energy was determined using the Tauc plot method. Photoluminescence(PL)spectroscopy was employed to study the recombination of electron and hole layers.

Electrochemical attributes were then examined potentiostatically using the chronoamperometry method. All electrochemical measurements were conducted using Ag/AgCl as the reference electrode in the usual three-electrode system.

2.4 PFC Setup

The PFC configuration was set up using a home-made single chamber reactor containing the $\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$ or $\text{BiVO}_4/\text{Cu}_2\text{O}$ as photocathode and a WO_3/TiO_2 as photoanode (Bachri *et al.*, 2025). The electrolyte was a 0.3 M KOH solution containing crude palm oil mill effluent (POME) adjusted to a chemical oxygen demand (COD) concentration of 100 mgL⁻¹. System illumination was provided by a 3 W LED light source with a 410–420 nm wavelength to mimic the visible light condition. Gas chromatography (GC, Shimadzu GC-2014 fitted) was used to determine hydrogen production quantitatively. The electric current produced by the system was recorded by a digital multimeter in an external circuit.

3. Results and Discussion

3.1 Structural and Morphological Characterization

Figure 1 shows XRD patterns that reveal the crystal structure and phase purity of the produced photocathodes. According to the JCPDS card No. 05-0667 (Lv *et al.*, 2018), the $\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$ photocathode exhibited diffraction peaks that were similar to the (111), (200), and (220) facets of the cubic-phase Cu_2O at the 2 theta value (36.4°, 42.3°, and 61.3°). The (002) plane reflection of $\text{g-C}_3\text{N}_4$ which arises from the interlayer stacking of aromatic structures in the polymeric graphitic structure, was evidenced by a broad diffraction peak at roughly

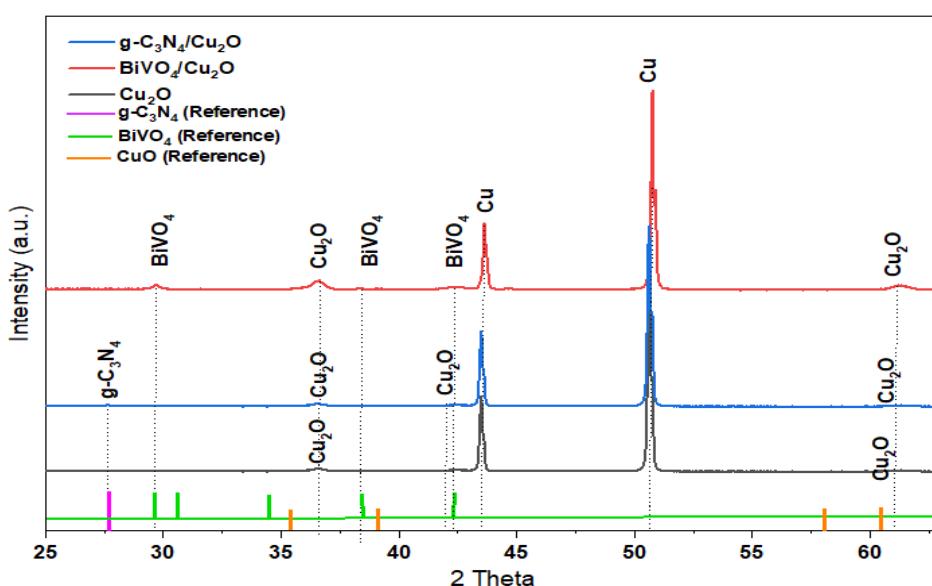


Fig. 1 Simulated XRD patterns of Cu_2O , $\text{Cu}_2\text{O}/\text{g-C}_3\text{N}_4$, and $\text{Cu}_2\text{O}/\text{BiVO}_4$ photocathodes (Source of $\text{g-C}_3\text{N}_4$: (Niaz *et al.*, 2023); BiVO_4 : (Cai *et al.*, 2023); CuO : (Uma *et al.*, 2021))

27.4°. The fact that the Cu_2O and $\text{g-C}_3\text{N}_4$ diffraction peaks were present together indicated that the p-n heterojunction structure also incorporated both components. (Zhang *et al.*, 2017; Cheng *et al.*, 2025).

The $\text{BiVO}_4/\text{Cu}_2\text{O}$ photocathode exhibited distinct peaks of Cu_2O , alongside additional peaks at $2\theta = 28.8^\circ$ and 47.2° , corresponding to the (121) and (040) planes of monoclinic scheelite-type BiVO_4 (JCPDS No. 14-0688 (Lin *et al.*, 2014)). The results demonstrate the development of a well-crystallized BiVO_4 layer on the Cu_2O surface by the immersion-calcination method. No impurity phases were searched in both samples, indicating the good phase purity and crystallinity of the two heterojunction photocathodes. The successful construction of $\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$ and $\text{BiVO}_4/\text{Cu}_2\text{O}$ heterostructures indeed offered a good basis for improved charge separation and the visible-light-driven photocatalytic performance, which was further validated by the investigations on the morphology, optical, and electrochemical properties in the following parts (Min *et al.*, 2014; (Hamdani & Bhaskarwar, 2021).

The crystallite sizes of the different phases in the $\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$ and $\text{BiVO}_4/\text{Cu}_2\text{O}$ photocathodes were estimated from the XRD patterns using the Scherrer equation (Table 1). For both composites, the Cu_2O (111) reflection appeared at $2\theta \approx 36.5^\circ$, with FWHM values of 0.40° for $\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$ and 0.37° for $\text{BiVO}_4/\text{Cu}_2\text{O}$, corresponded to crystallite sizes of 20.9 nm and 20.4 nm, respectively. This indicated that the modification

of Cu_2O with $\text{g-C}_3\text{N}_4$ or BiVO_4 did not significantly alter the Cu_2O crystallite size, and that nanosized Cu_2O domains (≈ 21 nm) were preserved in both photocathodes.

In the $\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$ sample, the (002) peak of $\text{g-C}_3\text{N}_4$ located at $2\theta = 27.44^\circ$ exhibited a relatively narrow FWHM of 0.11° , giving a crystallite size of 74.3 nm.

The larger crystallite size of $\text{g-C}_3\text{N}_4$ compared to Cu_2O suggested that $\text{g-C}_3\text{N}_4$ forms extended stacked domains that can act as a continuous light-absorbing and electron-transport layer in contact with the Cu_2O surface. In contrast, the monoclinic BiVO_4 phase in $\text{BiVO}_4/\text{Cu}_2\text{O}$, represented by the diffraction peak at $2\theta = 28.80^\circ$, showed a broader FWHM of 0.26° and a smaller crystallite size of 29.6 nm. The presence of BiVO_4 nanocrystallites in this size range implied a high interface density between BiVO_4 and Cu_2O , which is beneficial for constructing an efficient heterojunction.

These results demonstrated that both composites consist of nanoscale Cu_2O domains (~ 21 nm) intimately coupled with either larger $\text{g-C}_3\text{N}_4$ crystallites (~ 74 nm) or BiVO_4 nanocrystallites (~ 30 nm). Such size combinations are expected to provide a good balance between crystallinity (which favors charge transport) and interfacial area (which favors charge separation), thereby supporting the enhanced photoelectrochemical and photocatalytic performance of the $\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$ and $\text{BiVO}_4/\text{Cu}_2\text{O}$ photocathodes discussed in the following sections.

Table 1
XRD peak parameters and Scherrer crystallite sizes of Cu_2O , $\text{g-C}_3\text{N}_4$ and BiVO_4 phases in the composite photocathodes.

Sample	Phase	Peak (hkl)	2θ (°)	θ (°)	θ Rad	FWHM (°)	β (rad)	D (nm)
$\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$	Cu_2O	-111	36.490	18.245	0.318	0.4	0.007	20.904
	$\text{g-C}_3\text{N}_4$	-002	27.440	13.720	0.239	0.11	0.002	74.312
$\text{BiVO}_4/\text{Cu}_2\text{O}$	Cu_2O	-111	36.480	18.240	0.318	0.37	0.006	20.384
	BiVO_4	-121	28.800	14.400	0.251	0.26	0.005	29.583

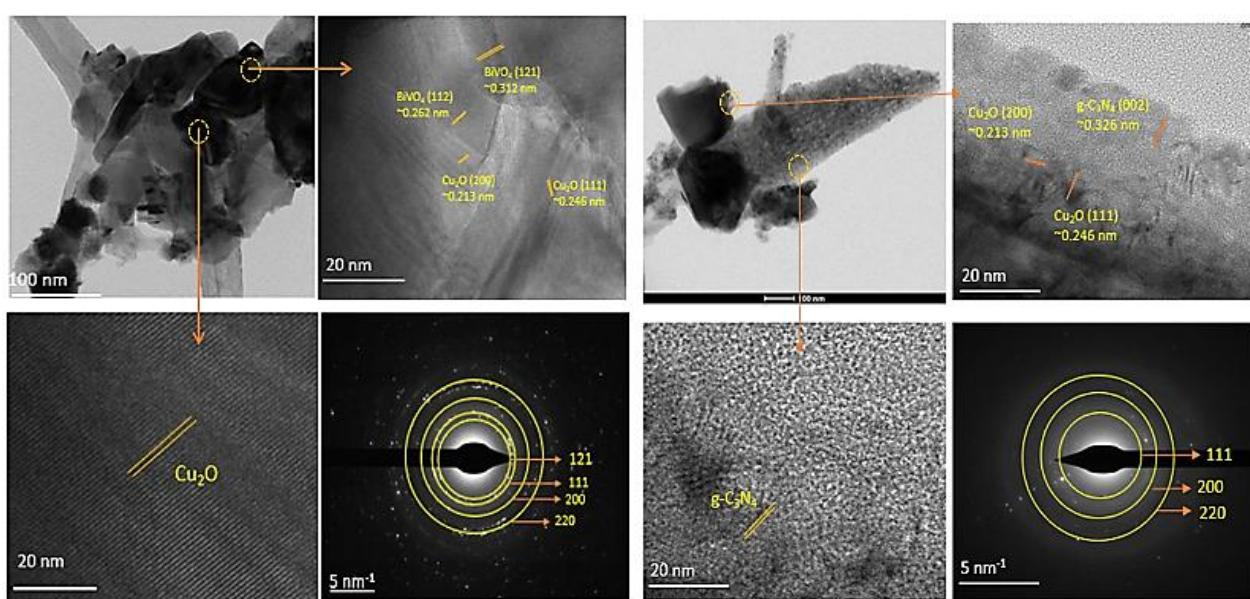


Fig. 2 The corresponding high-resolution TEM images, SAED patterns, and lattice fringes of (a). $\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$ (b). $\text{BiVO}_4/\text{Cu}_2\text{O}$ photocathodes

3.2 Surface Morphology Analysis

The surface morphology and crystallographic structure of the photocathodes were investigated by HRTEM (high-resolution transmission electron microscopy), SAED (selected area electron diffraction), and lattice fringe imaging as well (Figure 2). The overlaid $\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$ and $\text{BiVO}_4/\text{Cu}_2\text{O}$ layers were mainly well-shaped nanocubes, evenly adhering to the copper substrate (Kumar *et al.*, 2016). The nanocubes exhibited sharp edges and clear lattice fringes corresponding to the (111) and (200) planes of Cu_2O , indicating a highly oriented structure produced by anodization.

The as-obtained $\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$ composite (Figure 2), in which the Cu_2O nanocubes of about 40–90 nm in size were uniformly loaded onto the surface of $\text{g-C}_3\text{N}_4$ nanowires (Yang *et al.*, 2021), could be considered as a supporter framework. The bright and dark concave polygonal micro-structure of the $\text{g-C}_3\text{N}_4$ and Cu_2O phases suggested intimate interfacial contact at the boundary between them and demonstrated a good contrast difference between the two different components. Upon close examination, Cu_2O was still crystalline, whereas $\text{g-C}_3\text{N}_4$ seemed to be semi-amorphous, corresponding to the nature of its polymeric layer-like structure. Clear lattice fringes with an interplanar spacing of 0.246 nm were observed in the HRTEM picture (Figure 2), was in accordance with the (111) plane of the cubic Cu_2O (JCPDS 05-0667).

But $\text{g-C}_3\text{N}_4$ did not exhibit obvious fringes, indicating that it was semi-amorphous. Figure 2 also displayed an SAED pattern that indexed to a polycrystalline structure, with concentric diffraction rings that are indexed to the (111), (200), and (220) planes of Cu_2O . Figure 2 showed well-defined rings in the SAED pattern that correspond to the (111), (200), and (220) Cu_2O planes. It is expected that the Cu_2O nanocubes and the $\text{g-C}_3\text{N}_4$ nanowires are in strong contact, increasing the interfacial area between them. This increased interfacial area is expected to facilitate charge transfer and suppress electron–hole recombination.

On the contrary, there was a different interfacial configuration in the $\text{BiVO}_4/\text{Cu}_2\text{O}$ photocathode. BiVO_4

nanoparticles were preferentially attached at the junctions between Cu_2O nanocubes (interparticle-bridging network, not full conformal coating) (A. Wang *et al.*, 2015). This architecture provided the effective contact points between neighboring Cu_2O domains, and is expected to promote the directionality of the charge transport and the surface passivation. The HRTEM analysis showed lattice fringes from BiVO_4 with d spacings corresponding to (121) and (040) planes, while SAED patterns demonstrated a set of polycrystalline rings, suggesting well-crystalline BiVO_4 features in close proximity with Cu_2O .

3.3 Optical Properties and Band Gap Analysis

As shown in Figure 3, the photocathodes made of Cu_2O , $\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$, and $\text{BiVO}_4/\text{Cu}_2\text{O}$ have UV-Vis diffuse reflectance spectra (DRS). The absorption edge of the bare Cu_2O film was clearly approximately 620 nm, or 2.48 eV, in agreement with the characteristic of p-type Cu_2O semiconductors on the market. (Y. Zhang *et al.*, 2023). The graphitic carbon nitride π -conjugated structure and band gap of 2.7 eV are responsible for the increased absorption intensity in the 430–450 nm region seen in the $\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$ photocathode after doping with $\text{g-C}_3\text{N}_4$. (Alhaddad *et al.*, 2020).

The inclusion of $\text{g-C}_3\text{N}_4$ enhances absorption in the higher photon energy range with limited response to the long-wavelength visible light, as shown by the somewhat lower band gap of 2.51 eV, as demonstrated by the Tauc plot analysis. According to Cheng *et al.* (2025), BiVO_4 had a band gap of around 2.4 eV, whereas the $\text{BiVO}_4/\text{Cu}_2\text{O}$ photocathode showed a second absorption edge at 490 nm. In comparison to bulk Cu_2O and $\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$, the effective band gap calculated using the Tauc plot technique was 2.2 eV, suggesting that coupling with BiVO_4 may increase photon utilization under solar irradiation by expanding the absorption range into the longer visible light. (G. Li *et al.*, 2025). Using the Kubelka-Munk function, the optical band gap energy was determined using Tauc plots (J. Zhang *et al.*, 2024). According to the calculations, the bandgap energies of $\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$, Cu_2O , and BiVO_4 are 2.51 eV, 2.48 eV, and 2.2 eV, respectively.

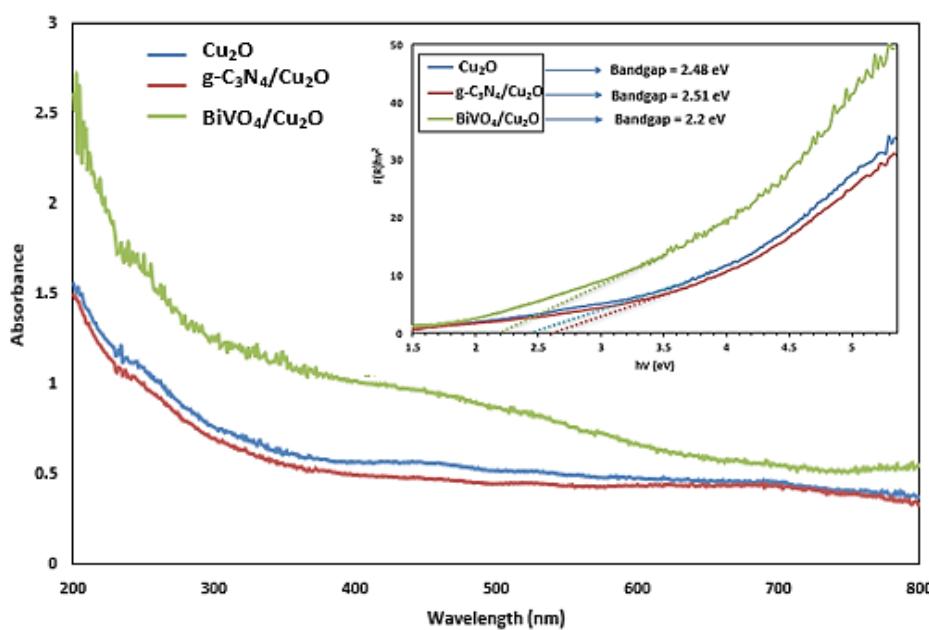


Fig. 3 UV-vis diffuse reflectance spectra (DRS) and Tauc plot analysis of Cu_2O , $\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$, and $\text{BiVO}_4/\text{Cu}_2\text{O}$ photocathodes

In both systems, the band edge alignments of the heterojunctions are expected to enhance the charge separation generated by visible light. The internal electric field is used to separate photogenerated carriers in the $\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$ system, which is achieved by creating a p-n heterojunction. (Anandan *et al.*, 2017). On the other hand, a type-II or Z-scheme structure may work better at the $\text{BiVO}_4/\text{Cu}_2\text{O}$ interface, allowing for efficient electron-hole transmission across the interface while minimizing recombination loss. (Cheng *et al.*, 2025). To improve the working performance in solar-driven hydrogen evolution and wastewater degradation in PFCs, the optical characteristics show that the two heterojunction photocathodes could absorb a broad spectrum of visible light.

3.4 Photoluminescence (PL) Analysis

In order to learn about the photocathode materials' photogenerated electron-hole pair recombination, steady-state PL spectroscopy was used. Under illumination at 325 nm, the photoluminescence (PL) emission spectra of the clean Cu_2O , $\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$, and $\text{BiVO}_4/\text{Cu}_2\text{O}$ are shown in Fig. 4. Figure 4 shows the photoluminescence (PL) spectra of the photocathodes. At around 630 a.u., the $\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$ composite exhibited the highest PL intensity of all the samples. The efficient recombination of photo-excited electron-hole pairs indicated by such high radiative recombination, which suggests that the interfacial contact between Cu_2O and $\text{g-C}_3\text{N}_4$ is not ideal for splitting charges.

The medium amount of carrier recombination was indicated by the weak PL emission with a peak intensity of around 420 a.u. from the as-prepared Cu_2O electrode. Despite being lower than $\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$, it indicated a limited carrier lifetime and poor charge-separation efficiency. Alternatively, the $\text{BiVO}_4/\text{Cu}_2\text{O}$ photocathode did not exhibit this property, and out of all the samples, its PL band had the least intensity at about 325 a.u. The photo-generated carriers in the sample indicated a much lower recombination rate and the most efficient charge separation is shown by the evident PL

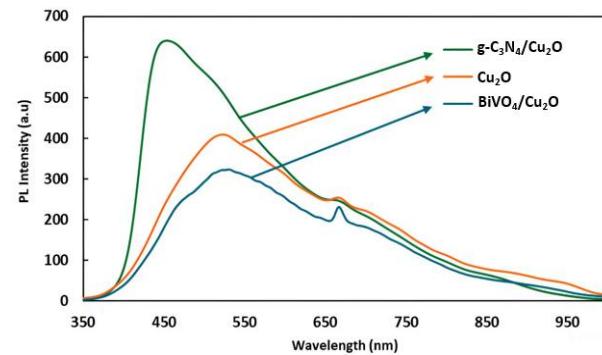


Fig. 4 Steady-state photoluminescence (PL) spectra of Cu_2O , $\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$, and $\text{BiVO}_4/\text{Cu}_2\text{O}$ photocathod

quenching. A type-II or Z-schemed heterojunction is essential for effective charge transfer across interfaces and for extending carrier lifetimes; the quenching of PL is a reaction to the advantageous band configuration between BiVO_4 and Cu_2O at the interface.

Furthermore, the FWHM values of the heterostructured samples were seen to be narrower in comparison to Cu_2O , indicating a little improvement in the spectra. This finding suggests that the charge transport energy level becomes more evenly distributed when the defect-related recombination center decreases (Min *et al.*, 2014). Collectively, the photoluminescence (PL) findings show that the $\text{BiVO}_4/\text{Cu}_2\text{O}$ heterojunction engineering dramatically improves photoelectrochemical performance and significantly reduces recombination losses.

3.5 X-ray Photoelectron Spectroscopy (XPS)

The surface composition and chemical states of the $\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$ and $\text{BiVO}_4/\text{Cu}_2\text{O}$ photocathodes were thoroughly investigated using X-ray photoelectron spectroscopy (XPS). The $\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$ system's Cu 2p spectrum showed two main peaks

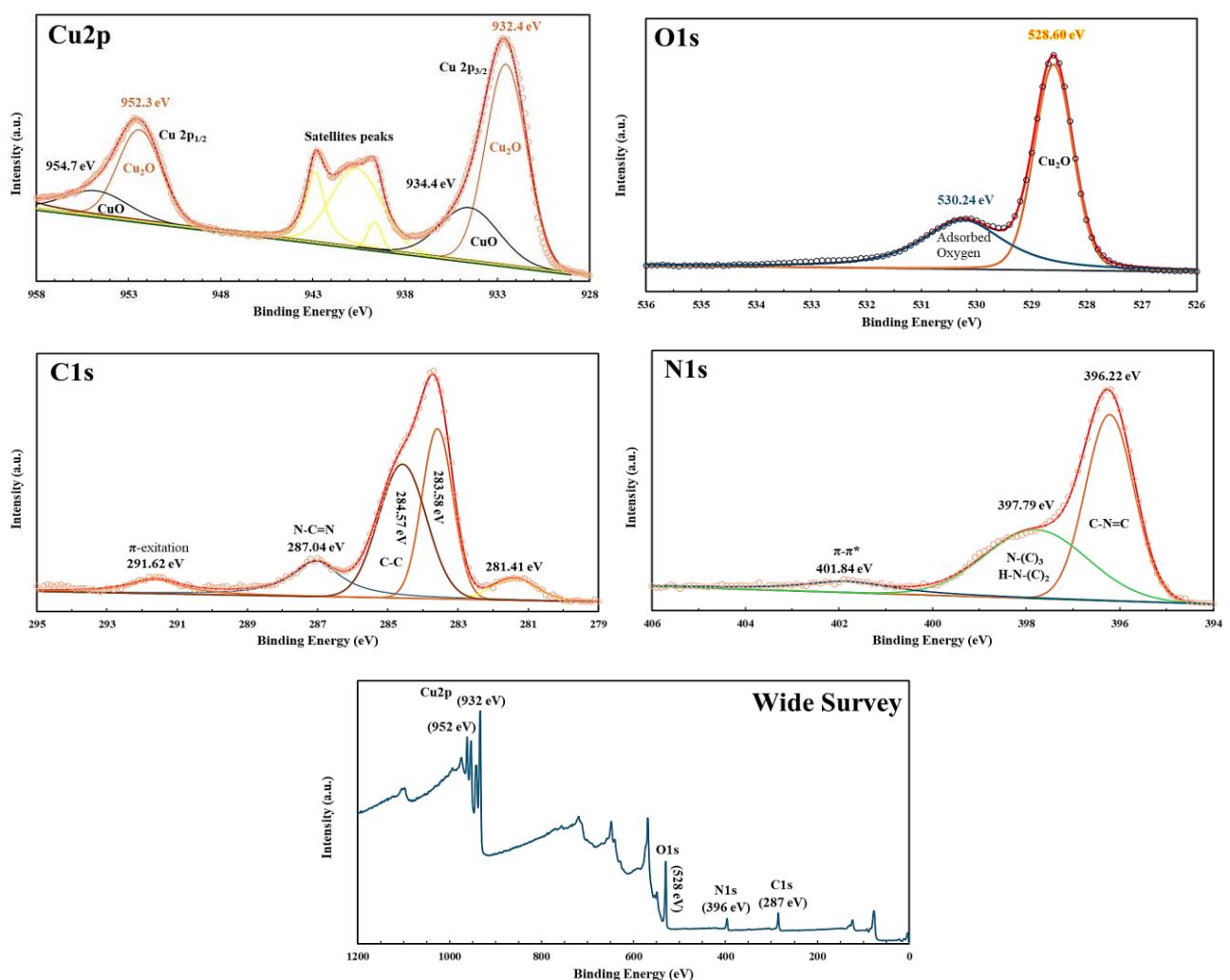


Fig. 5(a) XPS survey spectrum of $\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$ photocathode.

at 932.4 eV (Cu 2p 3/2) and 952.3 eV (Cu 2p 1/2), which was associated with Cu(I) species of Cu_2O . There were also shake-up satellite features between 941 and 945 eV, and a wider shoulder between 933 and 934.5 eV, which were indicative of the presence of some Cu(II) species, which are similar to CuO. The values observed in copper oxides are in agreement with the spin-orbit splitting of about 19.9 eV for Cu 2p_{1/2} and Cu 2p_{3/2}. Based on these findings, it seems that the photocathode is mostly an oxidation state of Cu(I) species in Cu_2O , with the presence of Cu(II) resulting from surface partial oxidation. Despite its 2p_{3/2} peak at 932 eV and lack of satellite characteristics, metallic Cu (Cu⁰) was not as noticeable as a significant component. Therefore, Cu_2O was the main component of the photocathode, whereas surface oxidation contributed to the remaining CuO.

There were two separated peaks in the O 1s peak, one at 528.60 eV and the other at 530.24 eV. The energy at 529.2 = 0.2 eV for Cu_2O , which is the lower binding energy (BE), was associated with lattice oxygen in the CuO phase (G. Li *et al.*, 2025), while the higher BE is thought to be caused by surface adsorbed oxygen or oxygen vacancies, the latter of which are active sites in photo-catalytic materials due to charge separation. According to Zeng *et al.* (2024), the photo-catalytic activity might be enhanced by using the adsorbed oxygen as electron traps.

In the C 1s spectrum, three peaks were resolved. The C-C bond, often caused by adventitious carbon or structural defect, was responsible for the main peak at 284.57 eV. The N=C-N bond coming from $\text{g-C}_3\text{N}_4$ (heptazine units) is corresponding to the second peak at about 287.04 eV. The aromatic system's $\pi-\pi^*$ shake-up transitions give rise to an additional satellite peak at 291.62 eV, suggesting a greater connection between the π -electron clouds in $\text{g-C}_3\text{N}_4$ as compared to ordinary C_3N_4 . At 396.22 eV, 397.79 eV, and 401.84 eV, there were three components to the N 1s spectra. The C-N = C bonds, which are characteristic of heptazine rings in $\text{g-C}_3\text{N}_4$, are thought to be responsible for the main peak at 396.22 eV (Y. Liu *et al.*, 2019). It is possible to attribute the second signal at 397.79 eV, which indicates bonding with three carbon atoms, to nitrogen in an N-(C)₃ or H-N-(C)₂ environment. The aromatic polymer chains' $\pi-\pi^*$ satellite nitrogen transitions were responsible for the high-binding energy peak seen at 401.84 eV.

The expected behavior for copper oxides was seen in the copper 2p region XPS spectra of the $\text{BiVO}_4/\text{Cu}_2\text{O}$ photocathode. At 932.4 eV, the primary peak emerged from Cu (I) species in Cu_2O ; at 933–934 eV, a much weaker shoulder emerged; and at 936–940 eV, shake-up satellite peaks emerged, which are ascribed to a small percentage of Cu(II) species (CuO-like). The findings showed that the Cu 2p_{1/2} peak was similar with the results for copper oxide (Ahmed *et al.*, 2020), with a

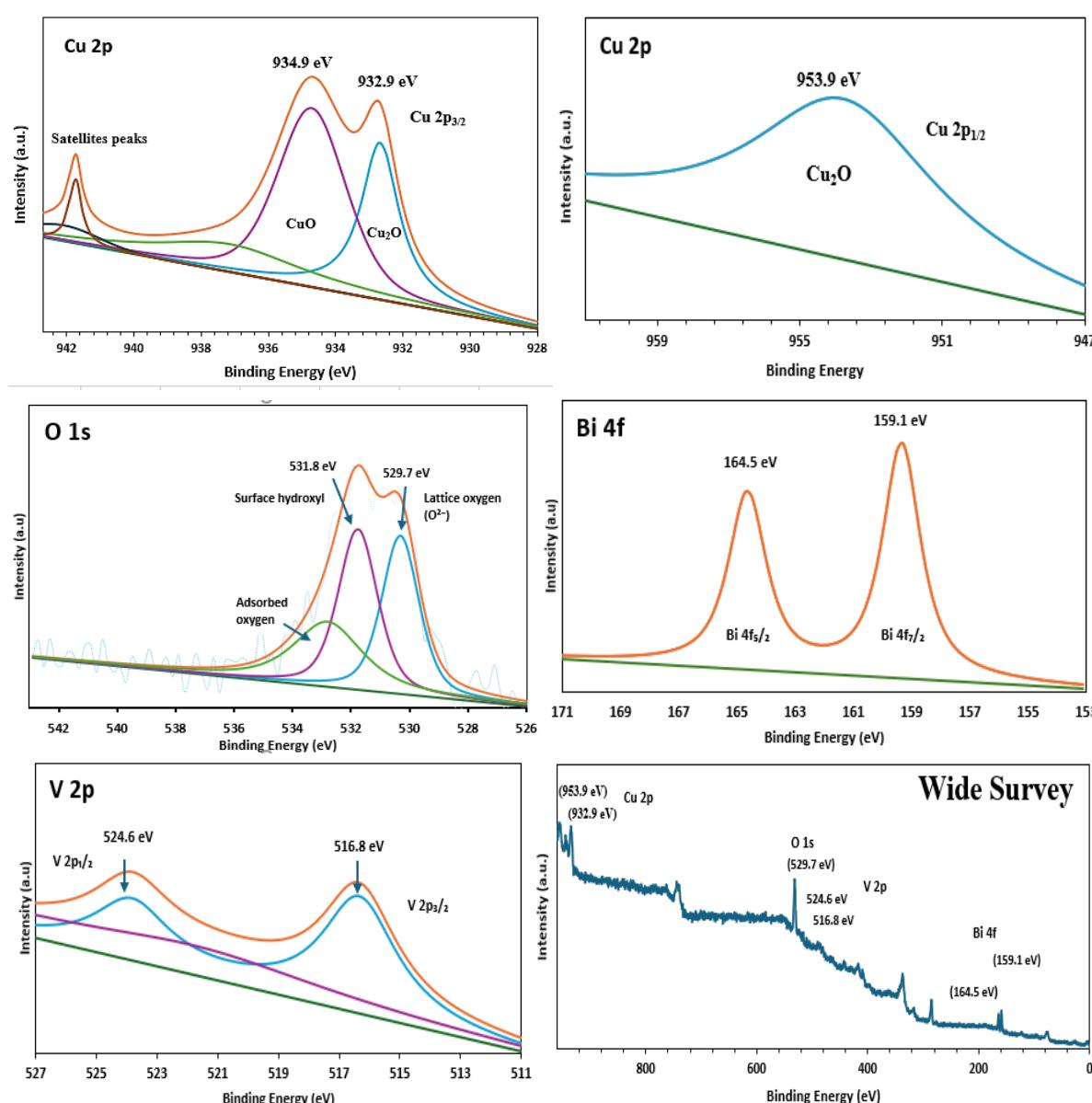


Fig. 5(b) XPS survey spectrum of $\text{BiVO}_4/\text{Cu}_2\text{O}$ photocathode

spin-orbit splitting of p 19.9 eV and a center at around 952.3 eV. Surprisingly, in the $\text{Cu } 2\text{p}_{1/2}$ area (955-962 eV), which is consistent with the intensity level of Cu(I) species, no extra satellite feature was seen. It has been confirmed by these results that Cu_2O is the most abundant copper species in the photocathode, with CuO exhibiting only a little amount, as is the case with most Cu-based photocatalysts. For photoelectrochemical activity and effective charge separation, a high Cu(I) concentration is ideal, whereas interfacial reactions might benefit from additional surface states offered by Cu(II) .

Consistent with BiVO_4 , the well-defined peaks of the Bi 4f region at 158.6 eV ($\text{Bi } 4\text{f}_{7/2}$) and 164.0 eV ($\text{Bi } 4\text{f}_{5/2}$) suggest the existence of the stable oxide Bi^{3+} . The absence of a reduction to V^{4+} , as shown by the peaks at 516.8 eV ($\text{V } 2\text{p}_{3/2}$) and 524.2 eV ($\text{V } 2\text{p}_{1/2}$) in V 2p, bodes poorly for the photocatalytic activities. Both the crystal lattice oxygen in VO_4 and Cu_2O and the surface hydroxyl groups or adsorbed oxygen on the bismuth vanadate surface were identified in the O 1s spectra of $\text{BiVO}_4/\text{Cu}_2\text{O}$, with the former peaking at 529.8 eV and the latter at 531.5 eV, respectively (Y. Zhang *et al.*, 2023). The photocatalytic activity

is enhanced by photogenerated electron trapping, and these oxygen-absorbed species are crucial for triggered interfacial redox processes. The heterojunction has the potential to improve the photocatalytic and photoelectrochemical capabilities, since each component has retained its physicochemical characteristics while the Cu_2O loaded with $\text{g-C}_3\text{N}_4$ and BiVO_4 worked well together.

High-resolution XPS (Table 2) confirms the coexistence of Cu^+ and Cu^{2+} species in both $\text{BiVO}_4/\text{Cu}_2\text{O}$ and $\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$ photocathodes. In $\text{BiVO}_4/\text{Cu}_2\text{O}$, Cu^+ in Cu_2O accounts for roughly 70% of the Cu 2p envelope, while surface Cu^{2+} (CuO -like) contributes the remaining ~30%, indicating that Cu_2O is still the dominant copper phase with only partial surface oxidation. The O 1s region is composed of lattice O^{2-} , surface hydroxyls, and adsorbed oxygen in comparable proportions, and the Bi 4f and V 2p spectra show single Bi^{3+} and V^{5+} doublets, confirming the presence of stoichiometric BiVO_4 in the composite. For $\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$, the relative fractions of Cu^+ (~72%) and Cu^{2+} (~28%) are similar, again suggesting a Cu_2O -rich bulk covered by a thin oxidized layer. The O 1s spectrum is

Table 2Deconvolution of high-resolution XPS spectra for Cu, O, Bi, V, C and N species on $\text{BiVO}_4/\text{Cu}_2\text{O}$ and $\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$ photocathodes.

Sample	Element	Core level	Main Assignment	Area	Area (%)		
$\text{BiVO}_4/\text{Cu}_2\text{O}$	Cu	$\text{Cu} 2\text{p}_{3/2}$	Cu^+ in Cu_2O	955.55	18.52		
			Cu^{2+} in CuO (surface)	1548.83	30.02		
		$\text{Cu} 2\text{p}_{1/2}$	Cu^+ in Cu_2O	2654.25	51.45		
$\text{BiVO}_4/\text{Cu}_2\text{O}$	O	O 1s	Lattice Oxygen	711.69	34.38		
			Surface Hydroxyl	763.14	36.86		
			Adsorbed Oxygen	595.30	28.76		
$\text{BiVO}_4/\text{Cu}_2\text{O}$	Bi	Bi 4f	Bi^{3+} in BiVO_4	1317.69	100.00		
	V	V 2p	V^{5+} in BiVO_4	645.72	100.00		
	Cu	$\text{Cu} 2\text{p}_{3/2}$	Cu^+ in Cu_2O	313000.90	48.85		
			Cu^{2+} in CuO (surface)	118919.40	18.56		
		$\text{Cu} 2\text{p}_{1/2}$	Cu^+ in Cu_2O	149142.80	23.28		
$\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$	O	O 1s	Cu^{2+} in CuO (surface)	59680.66	9.31		
			Lattice Oxygen	104196.00	59.92		
			Adsorbed Oxygen	69700.54	40.08		
$\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$	C	C 1s	$\pi-\pi^*$	3128.98	7.42		
			N-C≡N	7304.66	17.33		
			C-C / C=C	17691.80	41.97		
$\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$			C-M / carbide-like C (C-metal)	14555.10	34.53		
			carbide-like/strongly reduced C	2597.74	6.16		
			C-N=C	14892.09	51.35		
$\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$	N	N 1s	N-C ₍₃₎ , H-N-C ₍₂₎	11537.14	39.78		
			phi-phi*	2572.42	8.87		

dominated by lattice oxygen and adsorbed oxygen species, while the C 1s region reveals contributions from C-C/C=C, N-C≡N in the $\text{g-C}_3\text{N}_4$ framework, and a significant share of C-M (carbide-like) species, evidencing strong interfacial bonding between $\text{g-C}_3\text{N}_4$ and Cu_2O . The N 1s spectrum, characterized by C-N=C and higher-binding-energy N-(C)₃/H-N-C components together with a weak $\pi-\pi^*$ feature, further corroborates the formation of conjugated $\text{g-C}_3\text{N}_4$ domains. These surface chemical states support the existence of intimate heterojunctions that are favourable for charge separation in both composite photocathodes.

3.6 Photoelectrochemical Performance

The photoelectrochemical characteristics of the Cu_2O photoelectrodes were also studied in terms of amperometric I-t measurements under chopped visible-light illumination with a bias of -0.5 V vs Ag/AgCl (Figure 6). The as-synthesized Cu_2O electrode showed -0.025 mA/cm² steady-state photocurrent density with the lowest photocurrent response among all samples, due to poor charge separation and fast electron-hole recombination. The $\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$ photocathode presented a slight increase up to -0.035 mA/cm² of stable photocurrent density, indicating that a p-n junction between Cu_2O and $\text{g-C}_3\text{N}_4$ was constructed, which could effectively

facilitate charge carrier separation and transfer in comparison with CN-absent Cu_2O (Bae *et al.*, 2024).

In contrast, peak photocurrent density was achieved for $\text{BiVO}_4/\text{Cu}_2\text{O}$ at around -0.10 mA/cm² during light ON cycles

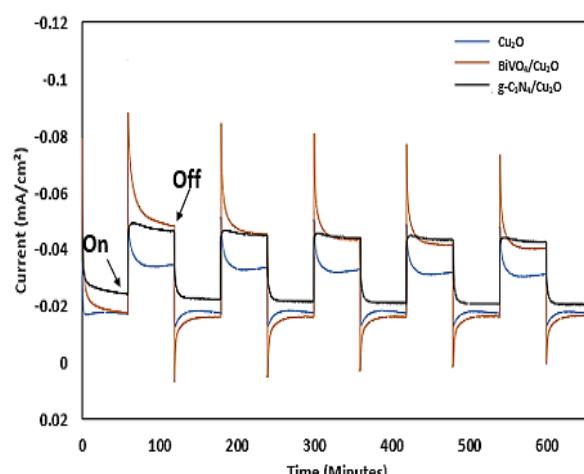


Fig. 6 Amperometric I-t curves of $\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$ and $\text{BiVO}_4/\text{Cu}_2\text{O}$ photocathodes at a bias of -0.5 V vs Ag/AgCl

and finally evolved to ~ -0.04 mA/cm 2 . This remarkable enhancement could be ascribed to the good band edge matching and efficient interfacial charge transfer, which result in the inhibition of recombination losses between BiVO₄ and Cu₂O (Shu *et al.*, 2025).

The repeated ON-OFF cycles confirmed the stable and reproducible photoresponse of all photocathodes, with BiVO₄/Cu₂O showing the strongest and most durable performance. These results clearly establish that incorporating BiVO₄ into Cu₂O was more effective in enhancing photocathodic activity than g-C₃N₄ modification.

3.7 Photocathode Stability

The stability trends shown in Figure 6 clearly distinguished the durability of the three photocathodes through their photocurrent behavior during repeated light on/off cycles. BiVO₄/Cu₂O consistently produced the highest photocurrent peaks, and the amplitude of these peaks remained largely preserved throughout the sequence, indicating minimal performance decay and highly efficient charge extraction under illumination. The slight downward drift observed after several cycles is modest compared to the other samples, demonstrating that BiVO₄ effectively mitigates the photocorrosion typically experienced by Cu₂O. Even after approximately four hours of operation, only faint localized darkening appeared on the surface, confirming that the structural integrity of the photocathode remains largely intact. In comparison, pristine Cu₂O showed moderate but relatively steady photocurrent responses, suggesting acceptable short-term stability but limited catalytic capability due to weak intrinsic charge separation. On the other hand, g-C₃N₄/Cu₂O exhibited the lowest photocurrent and the most pronounced cycle-to-cycle degradation, reflected by progressively diminishing peak intensities and visible surface deterioration such as darkening and roughened regions.

3.8 Analysis of Performance Testing for the PFC System

The stable and repeatable photoresponse for all photocathodes was verified using repeated ON-OFF cycles, where a BiVO₄/Cu₂O exhibited the highest and most durable performance. The results unambiguously demonstrate that BiVO₄ modification is much more effective in promoting the photocathodic activity than concurrent functionalization of g-C₃N₄.

3.8.1 Electricity Generation Performance

The current and power density characteristics, as shown in Figures 7(a) and 7(b), provide an electricity generation analysis. The electrical performance of the PFC cells indicated that the BiVO₄/Cu₂O photocathode bearing system was the most efficient compared to pristine Cu₂O, and the g-C₃N₄/Cu₂O photocathode showed the lowest reported performance. It was found that after 240 minutes of continuous photoelectrochemical operation, current and power densities of BiVO₄/Cu₂O reached around 0.0098 mA cm $^{-2}$ and 0.05625 mW cm $^{-2}$, respectively, which were higher than those of pristine Cu₂O (0.0076 mA cm $^{-2}$; 0.0375 mW cm $^{-2}$) and g-C₃N₄/Cu₂O (0.0052 mA cm $^{-2}$; 0.0260 mW cm $^{-2}$).

The BiVO₄/Cu₂O photoelectrode exhibited a significant enhancement in photocurrent and power generation, attributable to the interparticle bridging architecture of BiVO₄/Cu₂O, where BiVO₄ nanoparticles preferentially develop along Cu₂O nanocubes, forming conductive junctions that

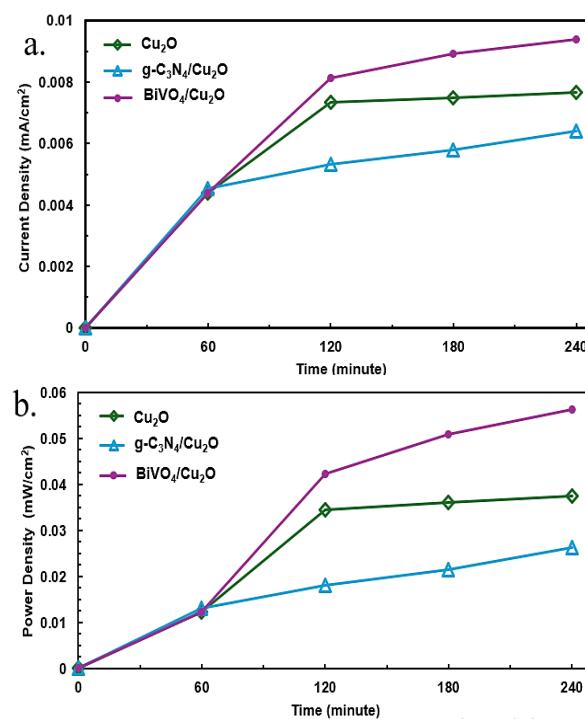


Fig. 7. (a) Current Density of Cu₂O, g-C₃N₄/Cu₂O and BiVO₄/Cu₂O photocathode 7(b) Power Density of Cu₂O, g-C₃N₄/Cu₂O and BiVO₄/Cu₂O photocathode

facilitate efficient directional charge transfer while minimizing electron-hole recombination. BiVO₄ may be illuminated by visible light, resulting in the transport of photogenerated electrons from the conduction band of BiVO₄ to the conduction band of Cu₂O, establishing advantageous energy levels (N. Li *et al.*, 2024). This synergistic electron transfer enhanced the electron availability for the hydrogen evolution process (HER), improved external circuit flow, and increases both HER efficiency and total power output. (Zhou *et al.*, 2025).

However, the electrical output of the g-C₃N₄/Cu₂O photocathode was lower than that of BiVO₄/Cu₂O and pristine Cu₂O, which can be ascribed to suboptimal interfacial matching and low mobility in the operating state within the nanowire network. These results also agree with previous reports on the critical importance of heterojunction engineering and interfacial optimization for photocurrent density and power output in PFC systems (Kee *et al.*, 2020; Slamet *et al.*, 2022; Pratiwi *et al.*, 2023).

3.8.2 Hydrogen Production Analysis

Figure 8 illustrates that the system based on the PFC with BiVO₄/Cu₂O photocathode produced the largest amount of hydrogen of 13.71 mmol m $^{-2}$ for cumulative volume, even for the longer time (240 min), which is slightly higher than that for the pure Cu₂O (11.19 mmol m $^{-2}$) or Cu₂O/g-C₃N₄ (8.16 mmol m $^{-2}$). The excellent HER performance of BiVO₄/Cu₂O can be ascribed to a unique interparticle bridging structure, since the BiVO₄ nanoparticles anchoring with Cu₂O nanocubes selectively to build conductive interparticle bridges further and approach the interface contact, promote directional charge transfer, and suppress the recombination of electron holes.

Under visible-light irradiation, BiVO₄ could absorb photons and quickly separate photoinduced electrons in its conduction band (CB), which were transferred to the CB of Cu₂O because of its well-engineered band structure. These electrons

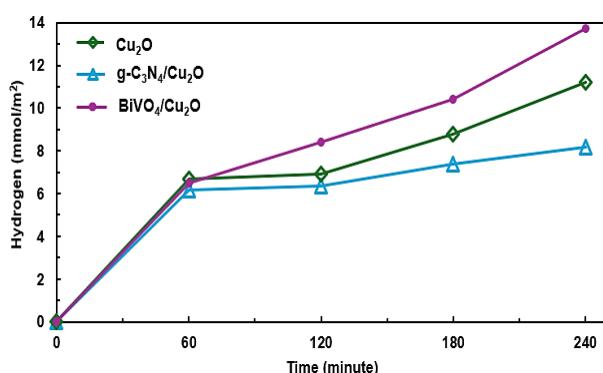


Fig. 8 Hydrogen evolution of Cu_2O , $\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$ and $\text{BiVO}_4/\text{Cu}_2\text{O}$ photocathode.

then further catalyzed HER at the surface of the photocathode. On the other hand, photogenerated holes in BiVO_4 could be transferred to the electrolyte and trigger oxidation reactions. The excellent hydrogen production suggested that the interfacial structure of $\text{BiVO}_4/\text{Cu}_2\text{O}$ suppressed the recombination in the photocatalyst and coordinates well the visible-light absorption, charge separation and surface reaction.

In contrast, $\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$ photocathode showed a relatively poorer hydrogen evolution activity, although the formation of a p-n junction between Cu_2O nanocubes and one-dimensional $\text{g-C}_3\text{N}_4$ nanowires was favorable for hydrogen evolution. The decreased hydrogen evolution was probably attributed to inefficient interfacial coupling and restricted charge mobility in the $\text{g-C}_3\text{N}_4$ network, thereby slowing down efficient electron transfer during operation.

These results agree with previous findings that rational heterojunction design and nanostructure engineering are two important strategies in order to enhance the hydrogen evolution performance in PFC systems (Husein *et al.*, 2024; Muttaqin *et al.*, 2022; Yu *et al.*, 2024).

3.8.3 Mechanism of Heterojunction in $\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$ and $\text{BiVO}_4/\text{Cu}_2\text{O}$ Photocathodes

The heterojunction formed between Cu_2O and its combined materials ($\text{g-C}_3\text{N}_4$ or BiVO_4) is instrumental in critically influences the overall performance of the PFC system (such as charge separation, electricity generation, and

hydrogen generation). According to the experimental result, it was believed that the $\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$ photocathode is dominated by the Type-II p-n heterojunction process (Chen *et al.*, 2014), and the $\text{BiVO}_4/\text{Cu}_2\text{O}$ photocathode has a more Z-scheme path (Khan *et al.*, 2022). They play a vital role in charge separation efficiency, recombination of electron-hole, and the redox potentials of active species (Ge *et al.*, 2019; Z. Wang *et al.*, 2021). To further substantiate these proposed charge-transfer pathways, electrochemical impedance spectroscopy (EIS) (Figure 9) was employed to probe the interfacial kinetics at the photocathode/electrolyte interface.

Electrochemical impedance spectroscopy (EIS) was carried out to gain further insight into the interfacial charge-transfer properties of the different photocathodes. The Nyquist plots in Fig. 9 exhibited a single depressed semicircle for all samples, characteristic of a charge-transfer process at the electrode/electrolyte interface that can be described by a simple Randles-type circuit. The diameter of the semicircle is directly related to the charge-transfer resistance (R_{ct}): a smaller semicircle reflects a lower R_{ct} and thus a faster interfacial charge-transfer rate. Fitting of the spectra yielded R_{ct} values of $\approx 8.0\text{ k}\Omega$ for $\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$, $\approx 3.0\text{ k}\Omega$ for bare Cu_2O , and only $\approx 0.5\text{ k}\Omega$ for $\text{BiVO}_4/\text{Cu}_2\text{O}$, while the solution resistances R_1 remained relatively small (≈ 50 , 90 , and $30\text{ }\Omega$, respectively).

A clear trend was observed in which $\text{BiVO}_4/\text{Cu}_2\text{O}$ shows the smallest semicircle and the lowest R_{ct} , followed by bare Cu_2O , whereas $\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$ displays the largest semicircle and the highest R_{ct} . This ordering ($\text{R}_{\text{ct, BiVO}_4/\text{Cu}_2\text{O}} < \text{R}_{\text{ct, Cu}_2\text{O}} < \text{R}_{\text{ct, g-C}_3\text{N}_4/\text{Cu}_2\text{O}}$) was fully consistent with the photoelectrochemical and photocatalytic performances, where $\text{BiVO}_4/\text{Cu}_2\text{O}$ outperforms Cu_2O and $\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$. The pronounced decrease in R_{ct} upon coupling Cu_2O with BiVO_4 indicated that the $\text{BiVO}_4/\text{Cu}_2\text{O}$ junction provided an additional, energetically favourable pathway for charge transport and effectively suppressed interfacial recombination. In contrast, the larger semicircle and high R_{ct} of $\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$ suggested a less efficient charge-transfer process, likely due to higher resistance across the $\text{g-C}_3\text{N}_4$ layer and/or less favourable band alignment. Taken together with the band-structure discussion, these EIS results strongly supported the formation of an efficient heterojunction in $\text{BiVO}_4/\text{Cu}_2\text{O}$, which facilitated directional charge separation and underpins its superior overall performance. When exposed to visible light, the $\text{Cu}_2\text{O}/\text{BiVO}_4$ Z-scheme heterojunction created photoinduced electrons (e^-) and holes (h^+). At the interparticle bridging contact, the valence

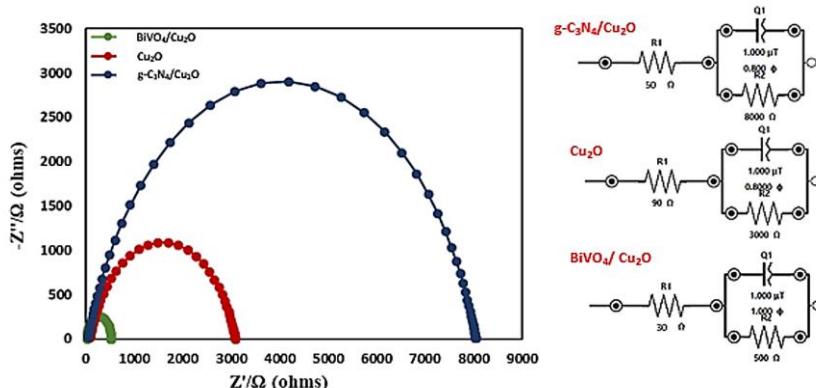


Fig 9. Nyquist plots and corresponding equivalent circuit models for Cu_2O , $\text{g-C}_3\text{N}_4/\text{Cu}_2\text{O}$ and $\text{BiVO}_4/\text{Cu}_2\text{O}$ photocathodes.

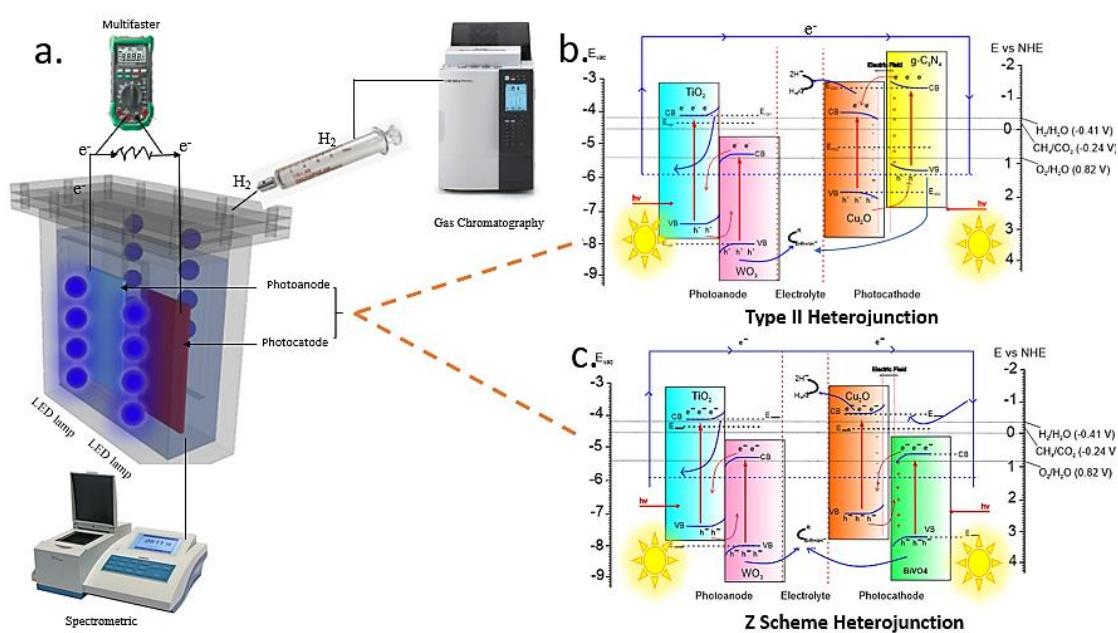


Fig. 10 (a) Predicted Schematic design of the photocatalysis-fuel cell reactor system with a WO₃/TiO₂-NT/Ti photoanode and g-C₃N₄/Cu₂O and BiVO₄/Cu₂O photocathode. (b) Prediction illustration of the charge transfer mechanism via a type-II heterojunction, and (c) Prediction via a Z-scheme heterojunction.

band holes of Cu₂O recombined with the conduction band electrons of BiVO₄. At the conduction band of Cu₂O, the reaction created electrons that were highly reducing, whereas at the valence band of BiVO₄, the reaction created holes that were very oxidizing. Consequently BiVO₄/Cu₂O performed very well in the simultaneous generation of electricity and hydrogen due to its better redox potentials, excellent interfacial charge transfer mechanism, and reduced bulk recombination.

The Type-II g-C₃N₄/Cu₂O heterojunction showed a staggered band alignment that moved electrons from g-C₃N₄ conduction band (CB) to Cu₂O CB (-1.1 V vs NHE) and holes from Cu₂O's valence band (VB) to g-C₃N₄ VB (+0.7 V vs NHE). The electron-hole recombination was effectively prevented by this spatial separation, although the redox potential was lower than in the Z-scheme system. As a result, when contrasted with BiVO₄/Cu₂O, the g-C₃N₄/Cu₂O photocathode produced less hydrogen and had lower electrical output.

The interface structure in both systems, together with the expected direction of charge migration and Fermi levels, is shown schematically in Figure 10. The emphasis is on the distinctions between the Type-II (g-C₃N₄/Cu₂O) and Z-scheme (Cu₂O/BiVO₄) processes. By selectively recombining low-energy carriers, the Z-scheme Cu₂O/BiVO₄ heterojunction might improve the exploitation of redox activity for PFC applications. On the other hand, the redox capacity is diminished in the Type-II g-C₃N₄/Cu₂O heterojunction, which primarily encourages spatial separation of the charges.

4. Conclusion

It is worth noting the different behaviors of Cu₂O-based photocathodes used in conjunction with g-C₃N₄ and BiVO₄ in the PFC electrodes. The BiVO₄/Cu₂O hybrid exhibited superior activity, delivering the highest hydrogen evolution rate (13.71 mmol m⁻²) and maximum power density (0.05625 mW cm⁻²) at 240 minutes, attributed to a proton gradient-driven Z-scheme heterojunction with a type-II interparticle bridging structure that promotes efficient charge transfer and suppresses carrier recombination. Direct Cu₂O displayed moderate activity (11.19

mmol m⁻², 0.0375 mW cm⁻²), whereas g-C₃N₄/Cu₂O showed the lowest activity (8.16 mmol m⁻², 0.0260 mW cm⁻²), clearly reflecting unsatisfactory charge separation and low redox potentials. These results underscore that interface optimization and band alignment are critical, and position the BiVO₄/Cu₂O-based photocathode as one of the most promising candidates for PFC devices for hydrogen production and power generation. It should be noted, however, that the assignment of Type-II (g-C₃N₄/Cu₂O) and Z-scheme (BiVO₄/Cu₂O) mechanisms remains largely inferential from photoelectrochemical, XPS, and EIS data, and future studies employing advanced spectroscopies such as ESR, transient photovoltage/photocurrent, or transient absorption will be required to directly verify the charge-transfer pathways.

The novelty of this work is twofold. First, a flat Cu plate was directly converted into Cu₂O and subsequently modified into two distinct heterojunctions, affording stronger interfacial contact, shorter electron pathways, and a more facile fabrication route than conventional FTO, ITO, carbon cloth, or Cu foam substrates. Second, both heterojunctions were integrated into a single-chamber PFC that simultaneously generated hydrogen and electricity, enabling a direct comparison of Cu₂O-based heterostructures under identical conditions. These outcomes highlight the importance of interface engineering and firmly establish BiVO₄/Cu₂O as a promising photocathode for future PFC applications.

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Author Contributions M.F.H.: Principal investigator responsible for data collection and primary data analysis. M.F.B.: Led the research, made methods and initial data processing. R.: Contributed to manuscript writing and helped with methodology, material collection, and purchase. R.Y.: Carried out materials characterization and participated in data interpretation. M.I.: Analysis and processing of photoelectrochemical properties data S.: Conceived the original idea and research plans, guided methodology, reviewed and revised the manuscript. All authors have read and agreed to the final version of the manuscript.

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