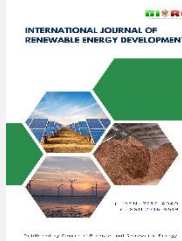




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Research Article

Hybrid precursor engineering of g-C₃N₄ toward improved α-Fe₂O₃/g-C₃N₄ photoanodes under HMF hole-scavenging conditions

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Abstract. The sluggish kinetics of Oxygen Evolution Reaction (OER) is a challenge for the development of efficient photoelectrochemical (PEC) systems for sustainable hydrogen production. In this work, α-Fe₂O₃/g-C₃N₄ photoanodes were prepared from g-C₃N₄ of different precursors (urea, melamine and/or dicyandiamide) to investigate the influence of hybrid precursor engineering on the photoelectrochemical properties of the heterostructures. The incorporation of hybrid precursor derived g-C₃N₄ modified the structural and optical properties of the composite photoanodes and resulted in enhanced PEC response compared to single precursor systems. The as prepared composite with g-C₃N₄ prepared from ternary hybrid precursors (urea, melamine and dicyandiamide) showed the best performance among the as prepared samples with lowest onset potential (0.01 V) and highest ΔE (0.88 V) under the present experimental conditions. Photoluminescence analysis indicated a lower emission intensity of the ternary hybrid precursor-derived sample, which was attributed to the inhibited charge carriers recombination. Furthermore, the addition of 5-hydroxymethylfurfural (HMF) as a model organic substrate increased the photocurrent density (~95%) with no change in the onset potential, demonstrating the effective hole scavenging ability of the HMF. Chronopotentiometry measurements confirmed stable operation during continuous illumination. These findings demonstrate that the hybrid precursor engineering of g-C₃N₄ can affect the PEC behavior of α-Fe₂O₃/g-C₃N₄ photoanodes under HMF-assisted conditions.

Keywords: α-Fe₂O₃ photoanode; g-C₃N₄ heterojunction; Hybrid precursor engineering; 5-hydroxymethylfurfural oxidation; Charge separation



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1. Introduction

Hydrogen has become a promising clean energy carrier for its high energy density (Møller *et al.*, 2017) and environmentally benign combustion product (Hossain Bhuiyan & Siddique, 2025). It is both a fuel for fuel cell technologies and an important feedstock for the chemical industry, especially for ammonia (AlZohbi, 2025; Bzovsky & Kudriavtsev, 2025), methanol (Hren *et al.*, 2024), and other value-added (Hren *et al.*, 2023) chemicals. But today, most hydrogen is still produced via steam methane reforming (Szablowski *et al.*, 2025), a process using fossil resources and producing huge amounts of CO₂ emissions, known as “gray hydrogen” (Dumančić *et al.*, 2024). These environmental concerns have led to significant interest in sustainable hydrogen production methods, such as water electrolysis powered by renewable energy (Al Dhahri *et al.*, 2026). Photoelectrochemical (PEC) water splitting is presented as a promising method for sustainable hydrogen production through the direct conversion of solar energy to chemical energy (Kamat & Sivula, 2022). In contrast to conventional electrolysis, PEC systems exploit the properties of semiconductor materials to absorb photons and produce electron-hole pairs, which trigger redox reactions at the

electrode surface. In this process, the semiconductor is directly involved in the conversion of light energy into electrical charges, where an internal electric field assists the separation of photogenerated electrons and holes (van de Krol, 2012). The semiconductor properties such as band gap, stability, cost, and photocurrent response are greatly affected by the efficiency of PEC systems (Zhu *et al.*, 2022). However, the oxygen evolution reaction (OER) at the photoanode is one of the main limitations in PEC water splitting because it demands high energy due to its sluggish kinetics, even though oxygen has a limited economic value (Corbin *et al.*, 2024). Several strategies have been developed to improve PEC efficiency in overcoming this limitation, which are based on reducing energy consumption and improving reaction kinetics (Chiang *et al.*, 2012). One promising approach is to replace the OER with alternative oxidation reactions that are thermodynamically more favorable, such as the oxidation of biomass-derived molecules, such as 5-hydroxymethylfurfural (HMF) (Luo *et al.*, 2021; Mumtazah *et al.*, 2026).

HMF is an important platform chemical derived from biomass, obtained through the conversion of simple sugars such as fructose and glucose. It is a precursor to an array of high value chemicals such as fuels, polymers and chemical intermediates

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(Arabi *et al.*, 2026; Zhang *et al.*, 2020). The use of HMF as an alternative oxidation substrate in PEC systems has advantages over water oxidation in terms of a lower overpotential and improved reaction thermodynamics (Luo *et al.*, 2021). Therefore, HMF-assisted PEC systems may improve charge utilization at the photoanode by facilitating hole consumption and suppressing competing OER pathways. The oxidation of HMF has been reported to be thermodynamically more favorable than OER. OER has a ΔG° of 237.2 kJ/mol and a redox potential of 1.23 V. The oxidation of HMF to 2,5-diformylfuran (DFF), however, has a much lower ΔG° of 178 kJ/mole (Ayed *et al.*, 2020; Liu *et al.*, 2021). Furthermore, DFF is a high value chemical with a market value far higher than that of HMF (Dubbink *et al.*, 2021). These advantages suggest that the replacement of OER with HMF-assisted oxidation processes may improve PEC performance while potentially enabling simultaneous value-added chemical conversion.

α -Fe₂O₃ (hematite) is one of the most studied semiconductor materials for PEC systems because of its abundance, low cost, non-toxicity and suitable band gap (~2.20 eV) (Zeng *et al.*, 2015). However, its performance is hindered by its poor electrical conductivity (Liu *et al.*, 2024), short hole diffusion length (Yu *et al.*, 2018), and not negative enough conduction band position for efficient hydrogen evolution. The limitations cause a rapid recombination of the electrons and holes, resulting in a greatly reduced photoelectrochemical efficiency. Therefore, some modification strategies are needed to enhance charge separation and surface reaction kinetics. A good strategy is by the addition of secondary semiconductors that enhance charge transport and catalytic activity. Graphitic carbon nitride (g-C₃N₄) has been considered as a promising modifier due to its chemical stability, visible light absorption, low cost, and relatively negative conduction band position (Ahmed *et al.*, 2024), which provides a strong reductive driving force for efficient electron transfer in photoelectrochemical systems (Pei *et al.*, 2023; Wang *et al.*, 2021). It can be prepared from nitrogen-rich precursors such as urea, melamine and dicyandiamide. The structural and electronic properties of g-C₃N₄, including the band gap, flat band potential, and charge transport properties, are strongly influenced by the precursor type. Various precursors have been reported to generate different morphology (Xia *et al.*, 2022) and photocurrent densities, e.g. 162.3 μ A/cm² (urea), 30 μ A/cm² (melamine), and 63 μ A/cm² (dicyandiamide) (Zhu *et al.*, 2022). The variations indicate the importance of precursor selection on the material performance.

Recently, it has been reported that the combination of multiple precursors can change the morphology and electronic structure of g-C₃N₄, which can affect the photocatalytic performance of g-C₃N₄ (Zhang *et al.*, 2019). The hybrid precursor systems have been reported to influence the porosity, surface area and electron transport properties that may affect the active site availability and charge carrier dynamics. Nevertheless, studies investigating hybrid precursor-derived g-C₃N₄ in α -Fe₂O₃/g-C₃N₄ PEC systems remain relatively limited. In this context, the coupling of α -Fe₂O₃ with g-C₃N₄ is a promising strategy to improve the photoanode performance. The introduction of g-C₃N₄ is expected to facilitate interfacial charge transfer through heterojunction formation (such as type-II or Z-scheme configurations) (Duan & Mei, 2020) rather than primarily improving visible-light absorption. Type-II heterojunctions may promote spatial separation of photogenerated charge carriers, while Z-scheme configurations can retain strong redox potentials during interfacial charge transfer. These interfacial charge transfer pathways are

complementary to the intrinsic photoactivity of α -Fe₂O₃, resulting in improved photoelectrochemical performance. The combination of both materials may influence charge carrier separation behavior and the overall photoelectrochemical activity (Ong *et al.*, 2016). Therefore, α -Fe₂O₃/g-C₃N₄ heterostructures are a promising strategy for low-cost and efficient PEC systems. The synthesis of α -Fe₂O₃ and α -Fe₂O₃/g-C₃N₄ composites has been reported in several studies. α -Fe₂O₃ has been synthesized by hydrothermal method (Kawde *et al.*, 2021) and α -Fe₂O₃/g-C₃N₄ composites have been prepared by hydrothermal synthesis followed by impregnation with various g-C₃N₄ loadings from urea precursors. Moreover, oxidation of HMF in PEC systems has been investigated with different photoanodes and mediators (Kawde *et al.*, 2021; Lhermitte *et al.*, 2021). Despite these advancements, there has been limited research on α -Fe₂O₃/g-C₃N₄ systems with hybrid precursor-derived g-C₃N₄ for PEC applications, particularly for systems involving HMF-assisted processes. Thus, this study investigates the influence of hybrid precursor-derived g-C₃N₄ on the structural, optical, and photoelectrochemical properties of α -Fe₂O₃/g-C₃N₄ photoanodes under HMF-assisted PEC conditions.

2. Material and Method

2.1. Materials

Fluorine-doped tin oxide (FTO) glass substrates were employed as the conductive support. Urea (CO(NH₂)₂, 99.5%, Merck), melamine (C₃H₆N₆, 99%, Merck) and dicyandiamide (C₂H₄N₄, 99%, Merck) were used as precursors for the synthesis of g-C₃N₄. Iron(III) chloride (FeCl₃, 97%, Sigma-Aldrich) and sodium nitrate (NaNO₃, 99%, Merck) were used as precursors for α -Fe₂O₃ synthesis. Ethanol (C₂H₅OH, 99.9%, Merck) and acetone (C₃H₆O, technical) were used as cleaning solvents. Electrolyte was prepared using potassium chloride (KCl, Merck, 99%). All experimental procedures were performed with deionized (DI) water. The electrolyte contained 5-Hydroxymethylfurfural (HMF, C₆H₆O₃, 99%, Sigma-Aldrich) as a model organic substrate. All chemicals were purchased from the respective suppliers and used without further purification.

2.2. Synthesis of α -Fe₂O₃ film

α -Fe₂O₃ thin films were deposited on fluorine-doped tin oxide (FTO) glass substrates by hydrothermal route. FTO substrates (10 mm × 50 mm) were cleaned with a mixture of ethanol and acetone (1:1, v/v) and rinsed with deionized (DI) water prior to synthesis. The precursor solution was prepared by dissolving 4.055 g of FeCl₃ and 8.50 g of NaNO₃ in 100 mL of DI water, with continuous stirring, until a homogeneous orange solution was obtained. Then, 60 mL of the prepared solution was transferred into a 100 mL Teflon-lined stainless-steel autoclave. The cleaned FTO substrate was then dipped in the solution with the conductive side down. The autoclave was sealed and kept at 195 °C for 6 h under its own pressure. After the reaction, the system was cooled to room temperature naturally. The obtained samples were removed, rinsed with acetone and dried in a nitrogen (N₂) stream. The substrates were washed with DI water and ultrasonicated for 5 min to remove loosely bound particles. Finally, the samples were dried for 2 h in an oven at 60 °C.

Table 1

Sample nomenclature and corresponding descriptions of $\alpha\text{-Fe}_2\text{O}_3$ and $\alpha\text{-Fe}_2\text{O}_3/\text{g-C}_3\text{N}_4$ photoanodes, along with $\text{g-C}_3\text{N}_4$ powders synthesized using single, binary, and ternary precursor combinations.

Sample Name	Description
Fe_2O_3	$\alpha\text{-Fe}_2\text{O}_3$ photoanode without the addition of $\text{g-C}_3\text{N}_4$
Fe-U	$\alpha\text{-Fe}_2\text{O}_3/\text{g-C}_3\text{N}_4$ photoanode with $\text{g-C}_3\text{N}_4$ prepared using urea precursor
Fe-D	$\alpha\text{-Fe}_2\text{O}_3/\text{g-C}_3\text{N}_4$ photoanode with $\text{g-C}_3\text{N}_4$ prepared using dicyandiamide precursor
Fe-M	$\alpha\text{-Fe}_2\text{O}_3/\text{g-C}_3\text{N}_4$ with $\text{g-C}_3\text{N}_4$ prepared using melamine precursor
Fe/U-D	$\alpha\text{-Fe}_2\text{O}_3/\text{g-C}_3\text{N}_4$ with $\text{g-C}_3\text{N}_4$ prepared using urea:dicyandiamide precursor (1:1)
Fe/U-M	$\alpha\text{-Fe}_2\text{O}_3/\text{g-C}_3\text{N}_4$ with $\text{g-C}_3\text{N}_4$ prepared using urea:melamine precursor (1:1)
Fe/M-D	$\alpha\text{-Fe}_2\text{O}_3/\text{g-C}_3\text{N}_4$ with $\text{g-C}_3\text{N}_4$ prepared using melamine:dicyandiamide precursor (1:1)
Fe/U-M-D	$\alpha\text{-Fe}_2\text{O}_3/\text{g-C}_3\text{N}_4$ with $\text{g-C}_3\text{N}_4$ prepared using urea:dicyandiamide:melamine precursor (1:1:1)
CN/U-D	$\text{g-C}_3\text{N}_4$ prepared using urea:dicyandiamide precursor ratio of 1:1
CN/U-M	$\text{g-C}_3\text{N}_4$ prepared using urea:melamine precursor ratio of 1:1
CN/M-D	$\text{g-C}_3\text{N}_4$ prepared using melamine:dicyandiamide precursor ratio of 1:1
CN/U-M-D	$\text{g-C}_3\text{N}_4$ prepared using urea:melamine:dicyandiamide precursor ratio of 1:1:1
CN/U	$\text{g-C}_3\text{N}_4$ prepared using urea precursor
CN/M	$\text{g-C}_3\text{N}_4$ prepared using melamine precursor
CN/D	$\text{g-C}_3\text{N}_4$ prepared using dicyandiamide precursor

2.3. $\text{g-C}_3\text{N}_4$ coating on $\alpha\text{-Fe}_2\text{O}_3$ for the synthesis of $\alpha\text{-Fe}_2\text{O}_3/\text{g-C}_3\text{N}_4$ film

Thermal polymerization of urea, melamine and/or dicyandiamide precursors with different compositions was used to synthesize $\text{g-C}_3\text{N}_4$ powder. A total of 20 g of precursor mixtures with different ratios (0:0:0, 1:0:0, 0:1:0, 0:0:1, 1:1:0, 1:0:1, 0:1:1, and 1:1:1) were mixed thoroughly before calcination. The mixtures were heated at 550 °C for 5 h to obtain $\text{g-C}_3\text{N}_4$ powder, then ground to obtain a fine and homogeneous material. For preparation of $\text{g-C}_3\text{N}_4$ dispersion, 100 mg of the obtained powder was dispersed in 100 mL of isopropyl alcohol. The suspension was ultrasonicated for 12 h to help exfoliation and then centrifuged at 3000 rpm for 10 min. The supernatant was then collected to a volume of 30 mL for further processing. The $\alpha\text{-Fe}_2\text{O}_3$ films were coated with $\text{g-C}_3\text{N}_4$ by means of an impregnation method. The films were soaked in the prepared $\text{g-C}_3\text{N}_4$ dispersion for 15 min followed by drying at 90 °C. This deposition-drying process was considered as one cycle and repeated 9 cycles for sufficient loading. The as-prepared $\alpha\text{-Fe}_2\text{O}_3/\text{g-C}_3\text{N}_4$ films were further calcined at 350 °C for 1 h and cooled down naturally to room temperature. The prepared $\alpha\text{-Fe}_2\text{O}_3/\text{g-C}_3\text{N}_4$ photoanodes were then used for further characterization and photoelectrochemical analysis, where the active electrode area was limited to 0.5 cm × 0.5 cm during PEC measurements.

2.4. Sample Nomenclature

In order to have a clear and consistent sample identification, a systematic nomenclature based on the precursor composition used for the $\text{g-C}_3\text{N}_4$ synthesis was employed. Abbreviations: U, urea; M, melamine; D, dicyandiamide. The notation "Fe/" indicates the use of $\alpha\text{-Fe}_2\text{O}_3$ as the base photoanode and the subsequent letters indicate the type and combination of precursors used to derive $\text{g-C}_3\text{N}_4$. Single, binary and ternary combination are indicated as such which allows for easy differentiation between samples. Table 1 gives a detailed description of all sample codes.

2.5. Characterization

The functional groups and bonding structures of the synthesized materials were studied by Fourier transform infrared (FTIR) spectroscopy using a Thermo Scientific Nicolet iS5 spectrometer. Spectra were taken in the 400-4000 cm^{-1} range with a DTGS KBr detector and KBr beamsplitter. All measurements were performed using 16 sample scans and 64

background scans to provide adequate signal quality. X-ray diffraction (XRD) analysis was performed to study the crystalline structure and phase composition of the samples by using a Rigaku Miniflex 600 diffractometer with Cu K α radiation ($\lambda = 1.5406 \text{ \AA}$) and a 1D D/teX Ultra detector. The measurements were performed in the 2θ range from 10 to 80° at a scanning rate of 10° min^{-1} . The optical properties of the fabricated photoanodes were measured by UV-Vis diffuse reflectance spectroscopy (DRS) performed on an Agilent Cary 60 UV-Vis spectrophotometer in the range of 200–800 nm. The energy of the band gap was calculated from the DRS spectra by the Tauc relation (Eq. 1):

$$(ah\nu)^n = A(h\nu - E_g) \quad (1)$$

where h is Planck's constant, ν is the photon frequency, α is the absorption coefficient and A is a proportionality constant. The exponent n depends on the type of the electronic transition. Indirect transition ($n=2$) was used to calculate the band gap energy of the $\alpha\text{-Fe}_2\text{O}_3/\text{g-C}_3\text{N}_4$ photoanodes, considering that the main phase in the composite (>90%) is $\alpha\text{-Fe}_2\text{O}_3$. To investigate the recombination behavior of the photogenerated charge carriers, photoluminescence (PL) analysis of the thin film samples was carried out. The emission spectra were recorded on a Horiba MicOS Photoluminescence, with an excitation wavelength of 325 nm and recording emission from 350 to 1100 nm.

2.6. Photoelectrochemical Measurement

Electrochemical measurements were performed in a typical three-electrode configuration with Pt mesh as the counter electrode, Ag/AgCl as the reference electrode, and $\alpha\text{-Fe}_2\text{O}_3/\text{g-C}_3\text{N}_4$ as the working electrode. The experiments were performed in electrolyte 1 M KCl under illumination of UV-visible light, with 30 W LED lamp as a source of light. The onset potential was determined from the LSV curves as the potential at which the photocurrent began to deviate noticeably from the baseline current. All potentials reported in this work were measured against an Ag/AgCl reference

3. Results and Discussion

3.1. Material Characterization

The functional groups and the bonding structure of the synthesized photoanodes were studied by FTIR spectroscopy

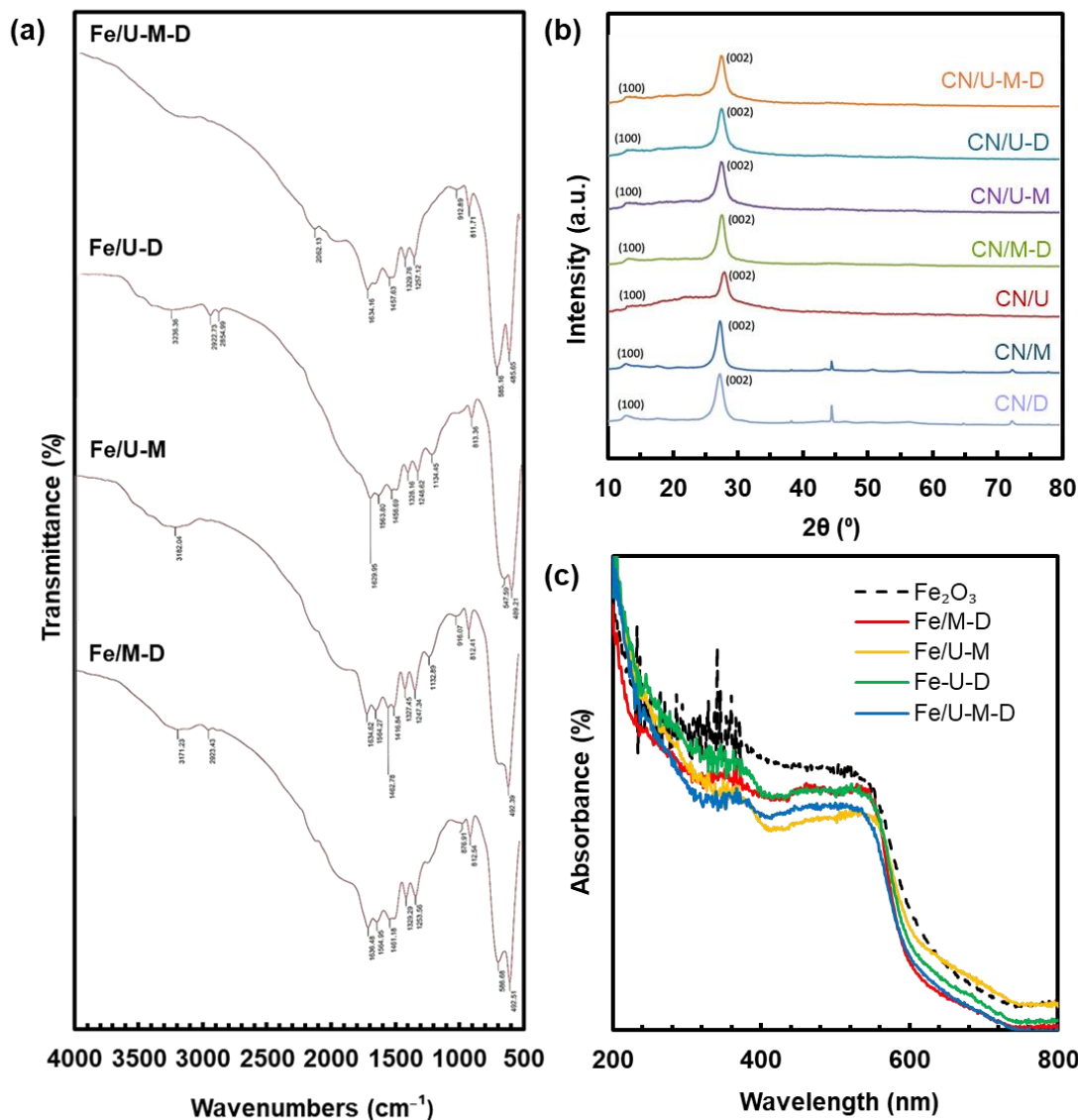


Fig 1. (a) FTIR spectra of α -Fe₂O₃/g-C₃N₄ (Fe/U-M-D) photoanodes; (b) XRD patterns of g-C₃N₄ synthesized from different hybrid precursor compositions; and (c) UV-vis spectra of α -Fe₂O₃/g-C₃N₄ photoanodes.

(Figure 1a). The α -Fe₂O₃ spectrum shows characteristic peaks at ~ 469 and 542 cm^{-1} attributed to Fe–O stretching vibrations. For g-C₃N₄, two different regions are observed: bands in the range 1210 – 1640 cm^{-1} assigned to C–N and C=N stretching of the aromatic framework (Ma *et al.*, 2017) and a sharp peak at $\sim 810\text{ cm}^{-1}$ assigned to the breathing mode of the heptazine units (Thaweesak *et al.*, 2017). The observed characteristic features in the composite samples (Fe/U-M, Fe/U-D, Fe/M-D, and Fe/U-M-D) indicate the presence of g-C₃N₄ within the α -Fe₂O₃-based composite films.

XRD patterns of g-C₃N₄ prepared with different compositions of hybrid precursors are shown in Figure 1b. The diffraction peaks at $\sim 13.1^\circ$ and 27.4° are attributed to the (100) and (002) planes of g-C₃N₄ (JCPDS No. 87-1526) (Zhao *et al.*, 2019) respectively, which are the evidence of in-plane structural ordering and interlayer stacking of the conjugated framework. The precursor composition has an effect on the crystallinity, where CN/M-D shows the highest peak intensity and higher degree of crystallinity than the urea-containing systems (CN/U-M, CN/U-D, and CN/U-M-D). This reduction in crystallinity is

probably associated with gas evolution during the decomposition of urea, which may disturbs the structural ordering during condensation process (Dong *et al.*, 2018). The crystallite size analysis (Table 2) demonstrates that hybrid precursor systems produce smaller crystallite sizes (13.05 – 17.79 nm) than g-C₃N₄ obtained from single precursor, with samples containing urea having the smallest sizes, consistent with previous studies (Ibadurrohman *et al.*, 2024). The reduced crystallite size may influence the surface characteristics and charge transfer behavior of the material, which could contribute to the observed photoelectrochemical response.

The optical properties of the photoanodes were investigated using UV-vis diffuse reflectance spectroscopy (Figure 1c). The absorption edges of Fe₂O₃, Fe/M-D, Fe/U-M, Fe/U-D, and Fe/U-M-D were observed at 689 , 639 , 663 , 654 , and 642 nm , respectively. The incorporation of g-C₃N₄ results in a slight blue shift in the absorption edge, accompanied by a marginal increase in band gap energy to 1.93 – 2.03 eV . This trend is consistent with the wider band gap of g-C₃N₄ relative to α -Fe₂O₃, indicating electronic structure modulation upon

Table 2
g-C3N4 Crystal Size with Hybrid Precursors.

Sample	2θ	FWHM	L (nm)
CN/D	27.24	0.81	32.21
CN/M	27.31	0.61	48.86
CN/U	27.81	1.09	34.28
CN/M-D	27.52	1.207	17.79
CN/U-M	27.47	1.349	15.09
CN/U-D	27.46	1.334	15.12
CN/U-M-D	27.38	1.406	13.05

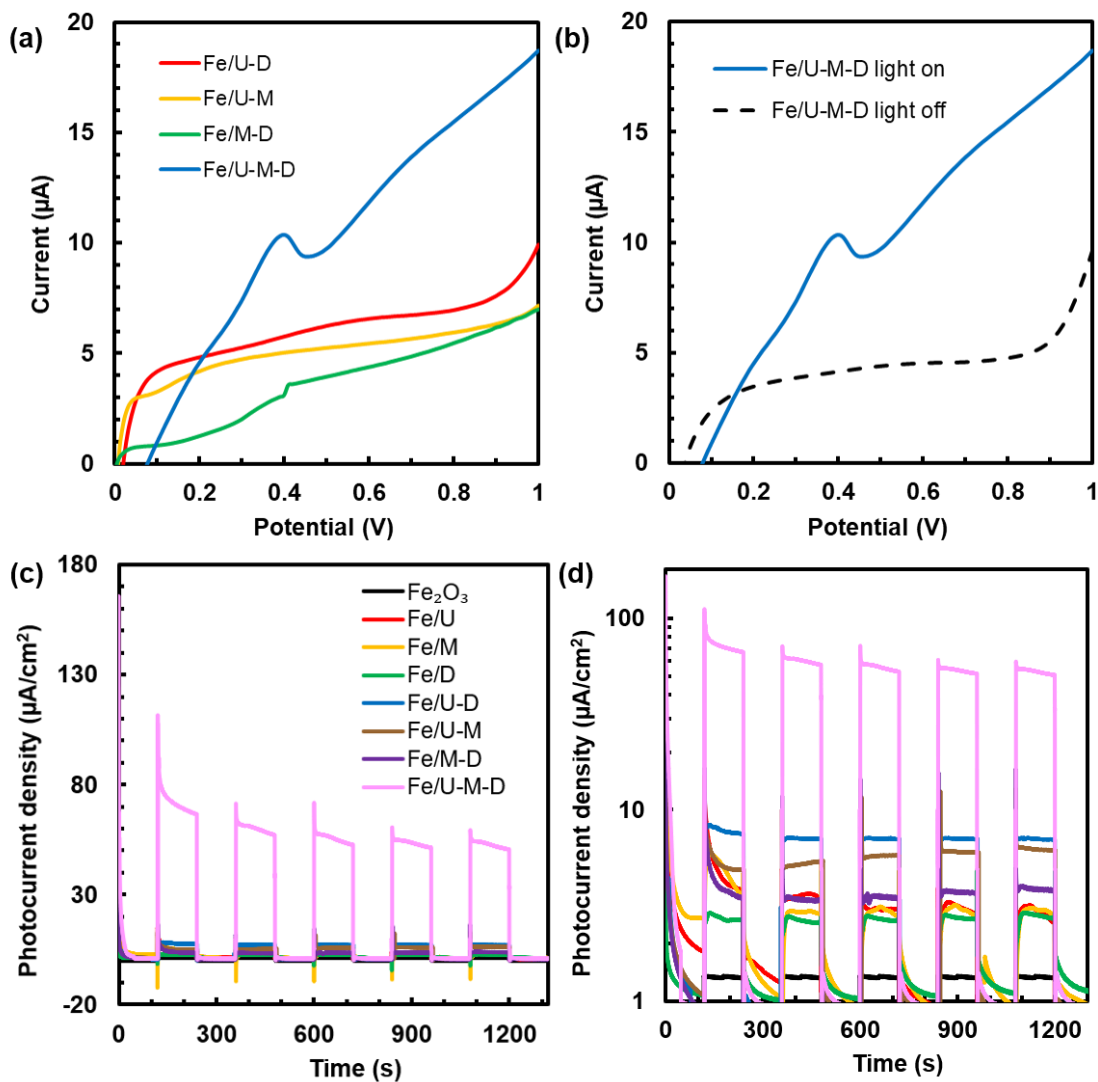


Figure 2. (a) Linear sweep voltammetry (LSV) curves of α -Fe₂O₃/g-C₃N₄ photoanodes with different hybrid precursor compositions under illumination (scan rate of 10 mV s⁻¹); (b) corresponding LSV curve of the best-performing sample (Fe/U-M-D); (c) chronoamperometry responses under intermittent light irradiation at 1.0 V (2 min dark–2 min light cycles, repeated five times); and (d) chronoamperometry data displayed on a logarithmic scale.

composite formation. Despite this shift, all samples remain active within the visible light region. More importantly, the incorporation of g-C₃N₄ may influence charge separation and interfacial charge transfer behavior through possible heterojunction interaction, which could contribute to the observed enhancement in photoelectrochemical performance.

3.2. Photo-electrochemical Characterization

Figure 2a presents the onset potentials of Fe/M-D, Fe/U-M, Fe/U-D, and Fe/U-M-D photoanodes under illumination, with comparison to dark conditions shown in Figure 2b. Under light irradiation, all photoanodes exhibit a noticeable shift in onset

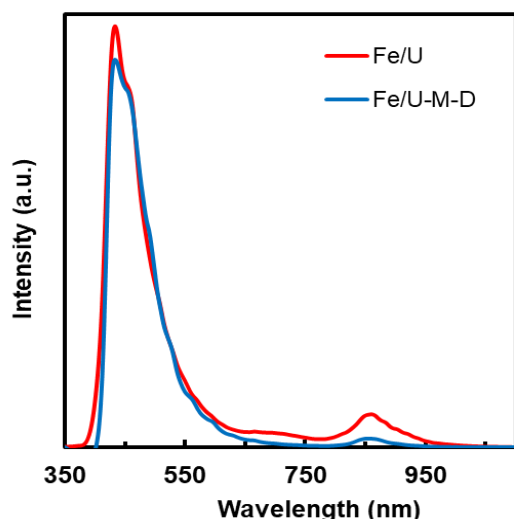


Fig 3. Photoluminescence (PL) spectra of $\alpha\text{-Fe}_2\text{O}_3/\text{g-C}_3\text{N}_4$ photoanodes synthesized using single precursor (Fe/U) and hybrid precursor (Fe/U-M-D), recorded at an excitation wavelength of 325 nm, illustrating reduced emission intensity for the hybrid system, indicative of suppressed electron-hole recombination.

potential toward lower values, indicating reduced energy requirements to initiate the reaction. This behavior is attributed to the generation of electron-hole pairs upon light absorption, which assists the electrochemical process and decreases the need for external bias. The difference in onset potential (ΔE) reflects the relative shift in onset potential under illumination. The onset potentials for Light Off, Fe/M-D, Fe/U-M, Fe/U-D, and Fe/U-M-D were determined to be 0.89, 0.03, 0.03, 0.07, and 0.01 V, respectively, corresponding to ΔE values of 0.00, 0.86, 0.86, 0.82, and 0.88 V. Among all samples, Fe/U-M-D exhibited the lowest onset potential and the highest ΔE under the present experimental conditions, indicating improved photoelectrochemical response under the tested conditions. In addition, the photocurrent density under illumination is significantly higher than that under dark conditions, suggesting improved photoresponse under illumination.

The negligible current observed in the absence of light further emphasizes the critical role of photoactivation in driving the reaction, demonstrating the important role of photoactivation in driving the electrochemical response. Chronoamperometry measurements were performed with intermittent light irradiation (2 min dark–2 min light cycles, repeated five times), as shown in Figure 2(c-d). The data in Figure 2d are presented on a logarithmic scale to improve visualization, whereas the original linear-scale data are provided in Figure 2c. The photocurrent densities of all composite photoanodes (Fe/M-D, Fe/U-M, Fe/U-D, and Fe/U-M-D) were higher than that of the pristine $\alpha\text{-Fe}_2\text{O}_3$, which suggests that the formation of $\alpha\text{-Fe}_2\text{O}_3/\text{g-C}_3\text{N}_4$ heterojunctions could enhance charge separation and transport. FTIR analysis also confirmed the presence of $\text{g-C}_3\text{N}_4$ by characteristic C-N-C and C-N heterocyclic vibrations. Transient photocurrent responses show an initial sharp increase upon illumination, which can be explained by the rapid photogeneration of charge carriers, followed by a decay to a steady state value due to recombination processes. The rate of decay indicates the competition between charge generation and recombination

(Mokhtarimehr & Tatarkova, 2017). In particular, Fe/U-M-D shows a more stable photocurrent response, suggesting the enhanced charge carrier dynamics. This conclusion is also supported by the photoluminescence analysis (Figure 3). The emission intensity of Fe/U-M-D is lower than that of Fe/U, indicating that the recombination of electron-hole pairs is suppressed and the charge separation is more effective, contributing to the improved photoelectrochemical performance.

3.3. Hybrid Precursor Effects

The improved photoelectrochemical activity of the samples prepared from hybrid precursors (urea, melamine, and dicyandiamide) shows a clear synergistic effect in the $\alpha\text{-Fe}_2\text{O}_3/\text{g-C}_3\text{N}_4$ system. This behavior agrees with previous studies on thiourea derived $\text{g-C}_3\text{N}_4$, where hybrid precursor routes have been reported to favor the formation of mixed triazine- and heptazine-based structures (Jing *et al.*, 2020). From a structural point of view, heptazine units have a more extended π -conjugated system than triazine, which allows for the delocalization of more electrons and more efficient creation and transport of charge carriers. The relevance of this mechanism to the present system can be understood from the similarity in the thermal decomposition pathways of urea and thiourea. The two precursors in the calcination process give cyanamide intermediates that polymerize to $\text{g-C}_3\text{N}_4$ frameworks. Such similarity in the hybrid systems with urea indicates the ability to induce similar structural motifs, which lead to better electronic properties. The experimental results show this structural effect, with Fe/U-D and Fe/U-M having higher photoelectrochemical activity than Fe/M-D and Fe/U-M-D having the highest performance of all. The improved activity of Fe/U-M-D may be associated with the combined influence of the different precursors on the structural and electronic properties of $\text{g-C}_3\text{N}_4$. Thus, the hybrid system has a smaller crystallite size (Table S1) which may influence the surface characteristics and charge transfer behavior of the material.

Meanwhile, the improved performance might be attributed to the interfacial charge transfer between $\alpha\text{-Fe}_2\text{O}_3$ and $\text{g-C}_3\text{N}_4$ through heterojunction construction. The more positive conduction band position of $\alpha\text{-Fe}_2\text{O}_3$ is less favorable for proton reduction, whereas $\text{g-C}_3\text{N}_4$ possesses a more negative conduction band potential and is thermodynamically more suitable for reduction reactions. Under illumination, the photogenerated charge carriers may be spatially separated across the heterojunction interface, which may result in suppressed charge recombination and improved photoelectrochemical response. This type of interfacial charge transfer may follow type-II or Z-scheme heterojunction behavior, as reported in previous studies (Lathe & palve, 2026). However, further band structure and charge transfer analyses are required to confirm the exact mechanism in the present system.

3.4. HMF-Assisted PEC Performance

In photoelectrochemical systems, the photoanodic reaction is typically limited by the oxygen evolution reaction (OER) that is energetically expensive and kinetically sluggish, but produces a low-value product (Luo *et al.*, 2021). To overcome this limitation, alternative oxidation pathways are often introduced using sacrificial agents that allow a more efficient hole consumption. Among these, the oxidation of 5-hydroxymethylfurfural (HMF) has been identified as a

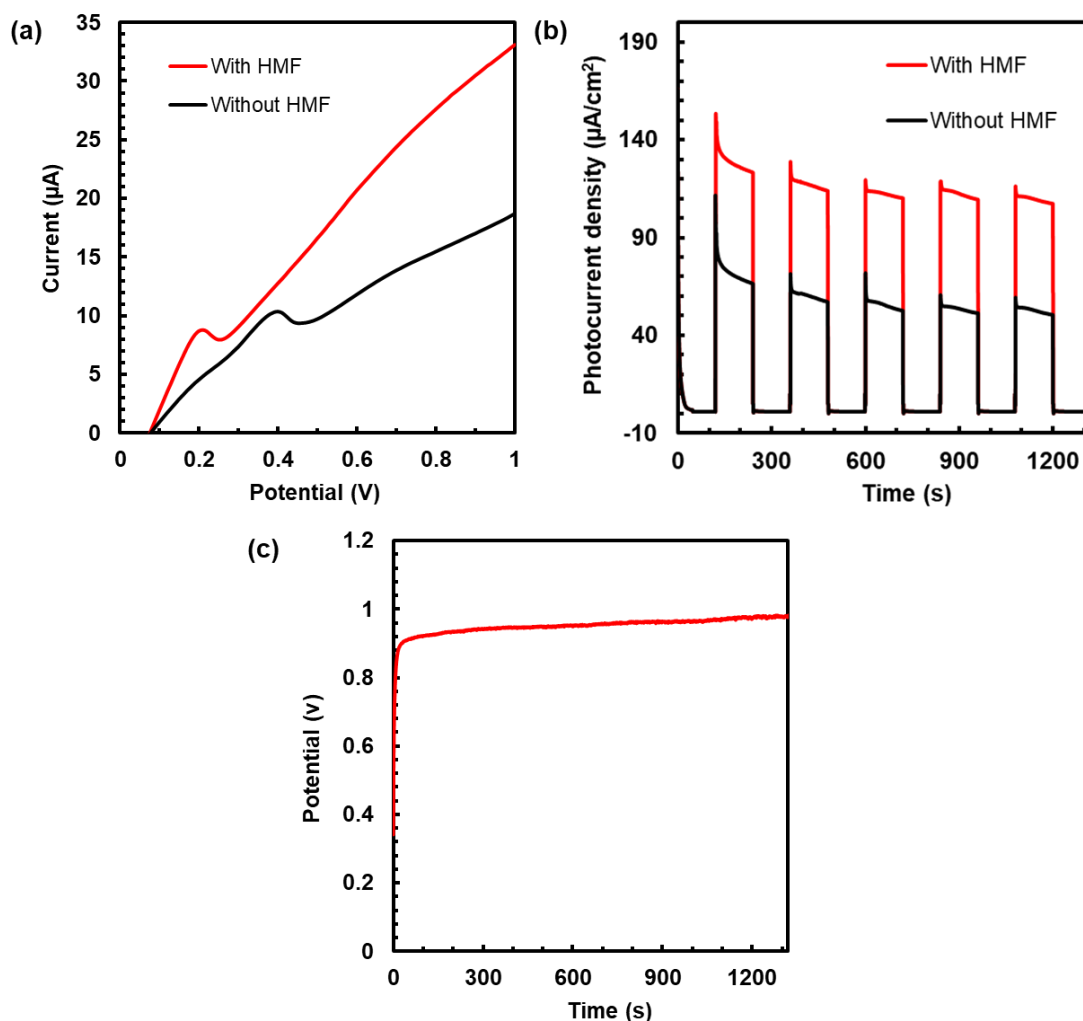


Fig 4. (a) Linear sweep voltammetry (LSV) and (b) chronoamperometry curves of Fe/U-M-D photoanodes with and without 10 mM HMF under illumination at 1.0 V; (c) Stability test using chronopotentiometry of Fe/U-M-D with 10 mM HMF at 30 μ A

promising alternative due to its faster kinetics and lower energy demand over OER. HMF was added to the electrolyte at a concentration of 10 mM in this study. The best performing photoanode, Fe/U-M-D, was then subjected to chronoamperometry measurements under the same cycling protocol (five cycles) to investigate the effect of HMF on the photoelectrochemical response.

LSV results show that the onset potential and ΔE are not affected by the inclusion of HMF in the electrolyte. As presented in Figure 4a, Fe/U-M-D and Fe/U-M-D with HMF have the same onset potential of 0.01 V and ΔE value of 0.88 V. However, a great increase in the current density is observed in the whole potential range in the presence of HMF. The enhanced current response indicates better charge carrier utilization where HMF acts as an efficient hole scavenger that results in faster consumption of holes and suppression of recombination. This behavior is in accordance with the chronoamperometry results, suggesting that the existence of HMF facilitates the photocurrent generation and improves the overall photoelectrochemical performance.

The photocurrent density of the Fe/U-M-D photoanode with and without 10 mM HMF at 1.0 V was 59.98 and 116.97 μ A cm^{-2} , respectively (Figure 4b). The presence of HMF led to a significant increase in photocurrent density of approximately 95%, indicating improved hole consumption during the

photoelectrochemical process. This behavior is associated with the role of HMF as a hole-accepting species that can compete with the oxygen evolution reaction and reduce charge carrier recombination. Chronopotentiometric measurements were conducted under continuous illumination for 2 h 40 min to evaluate the photo-response under continuous operation. Figure 4c shows that the measurement was performed at a constant current of 30 μ A, which was selected based on the photocurrent response observed from the LSV curve of the Fe/U-M-D photoanode in the presence of HMF. The potential rapidly stabilized and remained close to ~ 1 V throughout the measurement period, indicating relatively stable electrochemical behavior under the tested conditions. This potential response is consistent with the LSV results, where a photocurrent of ~ 30 μ A was obtained at ~ 1 V. Minor potential fluctuations were still observed during the measurement and may be attributed to interfacial charge transfer processes during continuous PEC operation.

4. Conclusion

α -Fe₂O₃/g-C₃N₄ photoanodes were successfully fabricated using g-C₃N₄ derived from ternary hybrid precursors (urea-melamine-dicyandiamide). Hybrid precursor engineering

influenced the crystallite size (13.05–17.79 nm) and optical properties, with absorption edges observed at 642–663 nm and band gap energies of 1.93–2.03 eV. Fe/U–M–D showed the best performance among all samples with an onset potential of 0.01 V, ΔE of 0.88 V and photocurrent density of 59.98 $\mu\text{A cm}^{-2}$. The addition of 10 mM HMF enhanced the photocurrent to 116.97 $\mu\text{A cm}^{-2}$ (~95% increase) without any change in onset potential, indicating its role as an effective hole scavenger under the tested conditions. Chrono-potentiometric measurements also showed stable photoresponse behavior during 2 h 40 min of continuous operation at ~1 V and 30 μA . Overall, these findings demonstrate that hybrid precursor-derived g-C₃N₄ can influence the photoelectrochemical behavior of $\alpha\text{-Fe}_2\text{O}_3/\text{g-C}_3\text{N}_4$ photoanodes under HMF-assisted PEC conditions.

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